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A GENERAL THREE-DIMENSIONAL COMPUTATIONAL MODEL FOR NONLINEAR C--ETC(U)

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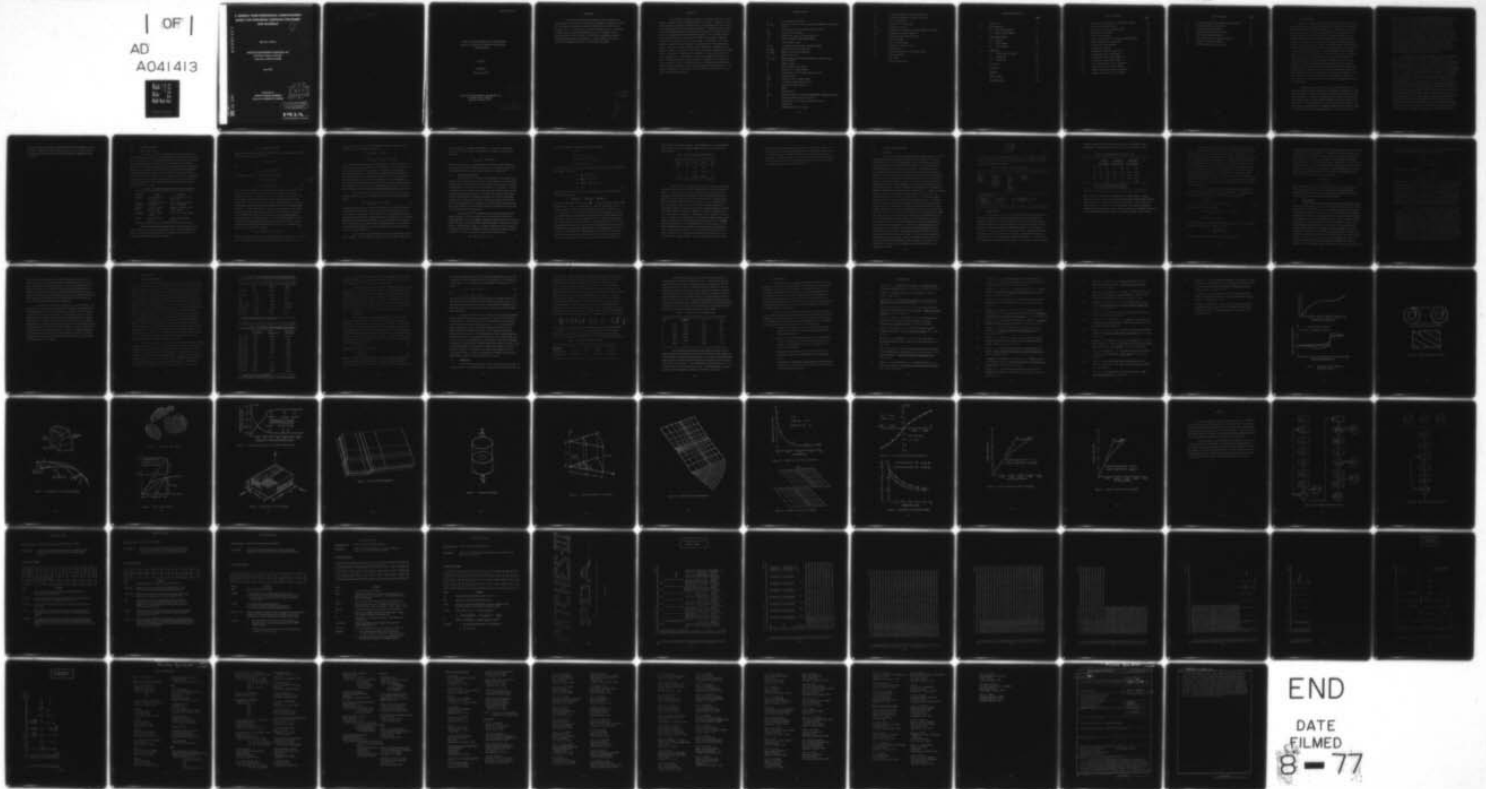
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**A GENERAL THREE-DIMENSIONAL COMPUTATIONAL
MODEL FOR NONLINEAR COMPOSITE STRUCTURES
AND MATERIALS**

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June 1977

**Prepared for
OFFICE OF NAVAL RESEARCH
Contract No. N00014-76-C-0161**

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FOREWORD

A general three-dimensional computational model for nonlinear composite structures and materials was developed by Prototype Development Associates, Inc. (PDA) under Contract Number N00014-76-C-0161 for the Office of Naval Research. Dr. E. L. Stanton served as Program Manager, Mr. L. M. Crain was responsible for scientific programming services, Dr. D. Mulville served as technical monitor for the Navy and Dr. J. Buch of PDA contributed a wealth of background information on material behavior. Their support and that of Dr. N. Perrone, who originated the project, are gratefully acknowledged.

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ABSTRACT

A computational model for the analysis of structural and microstructural behavior in general solids of composite material is presented. Emphasis is placed on representing the anomalous material behavior of composites and on the construction of computational models with variable properties. Alternative material models using continuum and statistical mechanics were reviewed and a modular code designed for compatibility with several different models. The constituent materials are characterized in terms of those state variables that correlate a materials response such as effective stress or strain energy. A parametric cubic representation is used for all state variables, the solid geometry and all physical properties. The associated finite element extends isoparametric modeling to allow properties, linear or nonlinear, to vary over the volume of an element as in rosette material construction. Applications of the model to a carbon-carbon unit cell, to strain singularities and to the inelastic response of a graphite bar illustrate its utility. Good agreement with triaxial test data for inelastic strains under hydrostatic pressure is obtained. Numerical results are computed using PATCHES-III and the conjugate gradient algorithm without the assembly of large matrices. This approach is tailored for vector processors and can reduce the high cost of nonlinear three-dimensional analyses.

NOMENCLATURE

a	Penny shaped crack radius
B_i, B_{ij}	Parametric cubic line and surface patch coefficients in geometric format
\tilde{B}	Binormal vector in the Frenet frame for a curve
C_{ijkl}	Stress-strain coefficients
ΔC_n	Difference in $C_{ijkl}(O, T)$ and $C_{ijkl}(V_n, T)$
c^0	Scalar multiplier in the Batdorf model
\tilde{e}_i	Cartesian basis vector
E_L, E_T	Longitudinal and transverse extensional moduli
F_{ij}, F_{ijkl}	Material strain state coefficients
$\bar{F}_{ij}, \bar{F}_{ijkl}$	Material stress state coefficients
\tilde{F}	Applied load vector
G_{LT}, G_{TT}	Shear moduli in the longitudinal-transverse plane and the transverse plane
$[K]$	Stiffness matrix
n	Index indicating cycle number
N	Dimension of a matrix problem
\tilde{N}	Normal vector in the Frenet frame for a curve
$P.E.$	Potential energy
$P(\xi)$	Parametric cubic property model
\tilde{P}	Conjugate gradient direction vector
\tilde{Q}_n	Pseudo-force vector at cycle n
r	Radius
\tilde{R}	Residual vector
S_i, S_{ij}	Parametric cubic line and patch coefficients in algebraic format
t_i	Conjugate gradient step size at cycle i
\tilde{T}	Tangent vector in the Frenet frame for a curve
T	Temperature
\tilde{u}	Displacement vector at a point

V_i	Variable representing a material strain state
\bar{V}_i	Variable representing a material stress state
W	Strain energy density
Z_i	Coordinate i in the e_i frame
$Z(\xi)$	Parametric cubic line
α_L, α_T	Longitudinal and transverse thermal expansion coefficient
α_{ij}	Coefficients of thermal expansion
β_i	Conjugate gradient direction factor
ϵ_{ij}	Strain components
θ	Polar coordinate angle
ξ_i	Parametric coordinate $0 \leq \xi_i \leq 1$
σ	Effective stress
σ_i^0	Normalizing stresses in the Batdorf model
σ_{ij}	Stress components
ϕ	Helix angle
w	Over-relaxation factor

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The use of composite materials for primary structure has led to demands for information on their three-dimensional response simply because uniaxial and biaxial loading often lead to important triaxial effects. This is true in the elastic range where edge effects can cause delamination and it is particularly true in the inelastic range where load redistribution can be strongly dependent on the microstructure of the material. One of the more readable essays on this subject is Drucker's¹ account in words and pictures of how different microstructures effect inelastic macroscopic response. Many of these involve inelastic mechanisms quite different from the crystallographic slip characteristic of metals. Some graphites exhibit biaxial softening due in part to microcracking² and several correlative models^{3,4} (i.e., phenomenological) are available that fit this data in the tension-tension quadrant. Quite the opposite behavior can occur in the compression-compression quadrant for another particulate composite, concrete, where the confinement of microcracks results in biaxial stiffening and again correlative models⁵ are available that fit this data. Anomalous inelastic behavior also occurs in fiber-reinforced composites as described in the survey paper by Francis and Bert⁶ who review the history of the fiber-matrix load transfer controversy. This issue is complicated by microstructural defects, particularly in composites subjected to severe processing conditions, by local fiber buckling and by residual processing strains.

Genuinely predictive inelastic models for composite materials are not available and most correlative models are an extrapolation of uniaxial data to multiaxial stress states. The situation in metals is similar⁷ but not to the same degree. Lin⁸ and Havner⁹, for example, have produced models that can predict multiaxial stress-strain response for polycrystalline aggregates and microstructural models at the atomic level¹⁰ (lattice statics) are also used in the material sciences. The microstructural detail required in a predictive material model for inelastic response clearly can be prohibitive for structural mechanics. In the case of composite

materials even the assumption of a statistically homogeneous material may not hold for certain behavior. However, when it does apply and when the principle of local action applies, we can use correlative models supported by adequate testing for structural mechanics. Operationally, this may require one or more pre-processors to obtain a correlative model and one or more post-processors to obtain results at the constituent level. Given this premise, the present effort focused on developing a computational system for all those correlative models that can be characterized by state variables. This is a very large class and will allow the numerical analysis of many composite structures whose inelastic behavior differs from metal plasticity. The approach also allows the model to function as its own pre-processor and synthesize correlative models from constituent inelastic properties and the behavior at interfaces.

The magnitude of the computational problem associated with inelastic three-dimensional analyses approaches that of characterizing the materials' physics. Several excellent survey papers are available¹¹ as well as case histories of recent applications.¹² The computational approach taken in the present paper is based on the premise that better computers are more likely than better numerical methods. Consequently, a computational method was adopted that takes advantage of what most scientific computers do best, vector processing. It is an adaption of the conjugate gradient algorithm that avoids connectivity optimization, that uses relatively little core and whose cycle time grows at most linearly. One of the method's disadvantages in linear problems, it is inefficient for multiple load conditions, can become an advantage in nonlinear problems when the iteration for the next load increment is started from the solution for the previous increment. A second aspect of the computational problem is that of data generation for the discrete mathematical model. This problem received considerable attention in the original development of PATCHES-III and resulted in a parametric cubic modeling system based on construction-in-context.¹³ The present effort added new geometry construction operations to the system, developed the parametrization necessary for modeling strain singularities, developed intra-element

property modeling and developed a common input format for all nonlinear material properties. Applications of these features are made to a carbon-carbon unit cell, to an interior crack problem and to the triaxial response of a graphite bar tested by Jortner.²

2.0 MATERIAL MODELS

2.1 State Variable Models

The objective is a general computational model that accounts for the anomalous inelastic behavior of composite materials, Table 1. The basic assumptions are that the material is statistically homogeneous and that at the macroscopic level inelastic behavior is strictly local. It is also tacitly assumed that this behavior is deterministic which may be the weakest link in the chain of assumptions. Physically based statistical models¹⁴ have been very successful in explaining the extensional behavior of graphites associated with the nucleation and growth of microcracks. However, additional work is required to include a shear mechanism and to reduce their computational expense before they can serve as the basis for a general computational model. Mathematically based statistical theories have not been nearly as successful.

Table 1 Anomalous inelastic behavior observed in composites

<u>Behavior</u>	<u>Metals</u>	<u>Composites</u>
Yielding	Independent of hydrostatic stress	Can be dependent on hydrostatic stress
Inelastic Mechanism	Crystallographic slip Twinning	Microcracking, interstitial slip, microbuckling
Inelastic Strain	Constant volume plastic strains	Inelastic strains can cause volume changes
Unloading	Occurs at elastic modulus	Can occur at secant modulus
Failure	Convex failure surfaces	Concave failure surfaces can occur

The approach taken assumes simply that the stress-strain law is some function of one or more stress or strain state variables. This assumption can be used for both incremental and total stress-strain laws but the present effort is limited to the latter and to three state variables

$$\sigma_{ij} = C_{ijkl}(V_1, V_2, T) \epsilon_{kl}^M \quad (1)$$

where one is temperature and ϵ_{kl}^M is the mechanical strain. Although not prescribed, commonly used V_1 and V_2 expressions are

$$V_1 = F_{ij}(V_1, V_2) \epsilon_{ij}^M$$

and

$$V_2 = F_{ijkl}(V_1, V_2) \epsilon_{ij}^M \epsilon_{kl}^M \quad (2)$$

with analogous expressions for strain-stress formulations

$$\begin{aligned} \bar{V}_1 &= \bar{F}_{ij}(\bar{V}_1, \bar{V}_2) \sigma_{ij} \\ \bar{V}_2 &= \bar{F}_{ijkl}(\bar{V}_1, \bar{V}_2) \sigma_{ij} \sigma_{kl} \end{aligned} \quad (3)$$

There is a formal similarity to the nonlinear stress-strain model of Nueber¹⁵, but his model restricts the form of Equation (1) and does allow the coefficients in Equations (2) and (3) to be independent functions of the state of the material. This is necessary to account for changes in the degree of anisotropy during inelastic straining.^{16, 17} A great many correlative models including Batdorf, Jones-Nelson, Weiler, Hahn-Tsai and the classical deformation models can be represented using Equation (1). It allowed the design of a nonlinear material module encompassing a variety of formulations all using one input format. All moduli and state functions are interpolated using piecewise cubics which are capable of approximating any continuous function as closely as required. This avoids the interpolation problems associated with exponential approximations in the Jones-Nelson model. The state variable used in this model is the strain energy density, $V_2 = W$, so that the F_{ijkl} are the C_{ijkl} and the nonlinear equation

$$W = C_{ijkl}(W) \epsilon_{ij}^M \epsilon_{kl}^M \quad (\text{Jones-Nelson}) \quad (4)$$

must be solved for each strain state. The Batdorf model is a strain-stress formulation that is very simple and accurate for transversely isotropic graphites. It uses an

effective stress parameter for a state variable that can be converted to a strain formulation by solving a nonlinear equation.

$$\bar{\sigma}^2 = \bar{F}_{ijkl} \sigma_{ij} \sigma_{kl} \quad (\text{Batdorf})$$

$$= \bar{F}_{ijkl} C_{ijmn}(\bar{\sigma}) C_{klst}(\bar{\sigma}) \epsilon_{mn}^M \epsilon_{st}^M \quad (5)$$

The moduli are linearly dependent on $\bar{\sigma}$ in this model. In order to account for tension-compression behavior differences, Batdorf uses an ad hoc procedure based on the sign of individual stress components. Unfortunately, this procedure, when used with Equation (5), often will lead to oscillating divergence between two stress states having the same $\bar{\sigma}$ but different signs. To avoid this, the signs are held constant during each major cycle in PATCHES-III. These difficulties are rewarded by surprisingly accurate agreement with the extensive biaxial and triaxial data obtained by Jortner.² Numerical results obtained using Equation (5) are compared with his data later in the paper for triaxial loading at constant strain ratios.

The Weiler model also uses an effective stress parameter for a state variable but allows the \bar{F}_{ijkl} to be dependent on the effective stress to account for changes

$$\bar{\sigma}^2 = \bar{F}_{ijkl}(\bar{\sigma}) \sigma_{ij} \sigma_{kl} \quad (\text{Weiler}) \quad (6)$$

in anisotropy with strain. The procedure used by Weiler and others¹⁸ is based on constraining the effective plastic work to be equal to the uniaxial plastic work in each component. Applications of these models have used \bar{F}_{ijkl} functions that make the effective stress insensitive to hydrostatic pressure, but this is not a requirement. Rybicki¹⁹ included plastic volume change in an \bar{F}_{ijkl} model for a carbon material, JTA, several years ago and obtained good correlation with data for a pressurized test cylinder under axial load. He, too, used an ad hoc procedure to account for tension compression differences.

The correlative models described have been used principally for particulate composites. It is also possible to represent most correlative models used

for fiber reinforced composites with Equation (1). Consider for example the Hahn-Tsai²⁰ model in which the nonlinearity for a laminar is shear dominated. In this case

$$V_2 = \bar{F}_{1212} \sigma_{12}^2 \quad (\text{Hahn-Tsai}) \quad (7)$$

and the extensional C_{ijkl} are independent of V_2 . There are essential differences between this model and those using strain energy for V_2 although at small strain levels both fit the data for graphite-epoxy laminates reasonably well.²¹

2.2 Spatial Variable Models

The objective is to model property distributions as accurately as deformations so that modeling detail is controlled by representation of the output rather than the input. This requirement is another instance in which composite material modeling differs from metals both for microstructures and structures. A carbon-carbon unit cell, for example, can be modeled with one variable property element or eight constant property elements. A helically wound structure with changing radii has properties that are continuously changing and is also most efficiently modeled with variable property elements. The approach taken in PATCHES-III is to use the same parametric cubic functions for both geometry and physical data. It is a generalization of the isoparametric approach and requires the development of new modeling techniques for properties.

Recently Henshell²² demonstrated how variable mesh point spacing (parametrization) can be used to greatly improve isoparametric models near a strain singularity. The same technique, changing the parametrization of the geometry model, can be used to model the step in material properties at an interior bimaterial interface. In this instance, Figure 1, we induce an inflection point in the property model at the interface by constructing a geometry model with an inflection point at the interface. Consider a coordinate function $Z(\xi)$ in algebraic format

$$Z(\xi) = S_1 \xi^3 + S_2 \xi^2 + S_3 \xi + S_4 \quad (8)$$

and impose the boundary conditions and inflection constraint

$$\begin{aligned}
 Z(0) &= S_4 \\
 Z(1) &= S_1 + S_2 + S_3 + S_4 \\
 \dot{Z}(\bar{\xi}) &= 0 = 3 S_1 \bar{\xi}^2 + 2 S_2 \bar{\xi} + S_3 \\
 \ddot{Z}(\bar{\xi}) &= 0 = 6 S_1 \bar{\xi} + 2 S_2
 \end{aligned} \tag{9}$$

where the step is at $\bar{Z} = Z(\bar{\xi})$. Next solve Equations (9) in terms of the as yet unknown $\bar{\xi}$ where $\Delta Z = Z(1) - Z(0)$.

$$\begin{aligned}
 S_1 &= \Delta Z / (3\bar{\xi}^2 - 3\bar{\xi} + 1) \\
 S_2 &= -3\Delta Z \bar{\xi} / (3\bar{\xi}^2 - 3\bar{\xi} + 1) \\
 S_3 &= 3\Delta Z \bar{\xi}^2 / (3\bar{\xi}^2 - 3\bar{\xi} + 1) \\
 S_4 &= Z(0)
 \end{aligned} \tag{10}$$

The location of the step or interface in parametric space, $\bar{\xi}$, can now be found by solving the cubic equation

$$\bar{\xi}^3 - 3 (\bar{\Delta Z} / \Delta Z) \bar{\xi}^2 + 3 (\bar{\Delta Z} / \Delta Z) \bar{\xi} - (\bar{\Delta Z} / \Delta Z) = 0 \tag{11}$$

for the root in the interval $0 \leq \bar{\xi} \leq 1$ where $\bar{\Delta Z} = \bar{Z} - Z(0)$ and Descartes' rule of signs guarantees there will be a root in the interval. Finally, a parametric cubic for a property component $P(\xi)$ is obtained by imposing the boundary conditions $P(0) = P_0$, $P(1) = P_1$ and $\dot{P}(0) = \dot{P}(1)$. Other choices are possible, for example, the value of $P(\xi)$ could be prescribed at four points, but this will cause $P(\xi)$ to fall outside the interval $[P_0, P_1]$ over portions of the interval $0 \leq \xi \leq 1$. This may be necessary if the area under the curve is a key parameter. One might keep the $\dot{P}(0) = \dot{P}(1) = 0$ constraint and choose the value of $P(0)$ such that the integral of $P(\xi)$ on the interval $0 \leq \xi \leq \bar{\xi}$ gives the correct area, $P_0 \bar{\Delta Z}$, with the corresponding choice for $P(1)$. In general, the determination of the parametrization and the intra-element property model form a nonlinear programming problem once an accuracy criteria such as least squares is established. However, a great deal can be accomplished with

simple analyses of the type presented. Parametrizations for a strain singularity at $Z(0)$, for example, can be obtained without solving a cubic and several are listed in Table 2.

Table 2 Strain singularity parametrizations

$Z(\xi)$	$Z(0)$	$Z(1/3)$	$Z(2/3)$	$Z(1)$
ξ^2	0	$L/9$	$4L/9$	L
ξ^3	0	$L/27$	$8L/27$	L
$\xi^2 + \xi^3$	0	$4L/27$	$20L/27$	L

Consider next the modeling of a fiber reinforced structure in which the orientation of the material axes changes continuously with respect to a reference frame. An interesting example is rosette construction used in rocket nozzle structures in which the fibers spiral about an axis with changing radii, Figure 2. Pagano²³ provides a detailed account of the relations between coordinate frames and shows that while axisymmetric, the properties vary in the radial direction. Also, all 21 elastic constants are non-zero for this material in cylindrical or rectangular coordinates making efficient property modeling particularly important. The transformation from material coordinates to cylindrical coordinates in this case is determined by a rotation $\varphi = \text{constant}$ followed by a rotation $\alpha = \alpha(r)$ where $r \sin \alpha = \text{constant}$. The transformation to rectangular coordinates simply requires adding a rotation θ to α . Intra-element property modeling in this case **requires** only the ply properties and the spatial distribution of the Euler angles. The original PATCHES-III system was designed for such input but did not anticipate the need for variable Euler angle data. As a result, the spatial variation of all 21 elastic constants had to be modeled individually using a pre-processor for a one-element model of a rosette cylinder. The same model could have been created

from the spatial variation of three Euler angles, two of which are constant. Comparisons between the parametric cubic properties and the analytic showed differences of less than one percent. Complete results from this study will be presented in a later paper. Finally, it should be remembered that variable property modeling adds no new degrees-of-freedom to the analysis model and can reduce the number of elements required for some composite materials.

3.0 COMPUTATIONAL MODELS

3.1 Geometry

The basic constructions and properties for parametric cubic line, surface and volume models may be found in the work of Coons²⁴ and others¹³. They provide a data base for computer aided geometric design comprehensive enough to be used for parts definition in manufacturing²⁵ and they now are used extensively in industry here and abroad to define external surfaces. Their reliability derives from modeling in parametric space where there are no problems with asymptotic slopes and their accuracy derives from using Hermite polynomials which interpolate a function and its derivatives with the smallest possible error. The price for these qualities is twelve coefficients for a line, forty-eight for a surface patch and one hundred ninety-two for a volume hyperpatch. The data generation problem for these coefficients can be solved using the construction-in-context approach developed originally for PATCHES-III. This system uses a variety of LINE___, PATCH___ and HP___ directives with cross referencing to construct the geometry model. Two construction operations, ruled volume and outline surface, are illustrated in Figure 3 and a listing of the currently available options is provided in Table 3. The outline surface construction operation shown in Figure 3 consists in having the computer move an outline curve along a base curve with a fixed orientation relative to either a global Cartesian frame or the local Frenet frame of the base curve. In developing this option, it became obvious that in many instances it would be desirable to change the initial orientation of the outline curve. This feature was provided by allowing an initial transformation that leaves the file copy of the outline curve unchanged. In retrospect, many of the original PATCHES-III directives would have benefited from such a feature which functions as a modifier in the language implicit to the present approach. The benefits of having even a primitive language have been substantial for data generation. The system functions as its own pre-processor and allows shapes, like those shown in Figure 4, to be created using roughly ten input directives (cards) per model. However, the models created contain several hundred coefficients that completely describe the geometry of the figure. The Frenet frame for a line $Z(\xi)$, for example, can be computed directly from the parametric cubic model

$$\begin{aligned}\tilde{T} &= \dot{\tilde{Z}} / |\dot{\tilde{Z}}| \\ \tilde{N} &= \ddot{\tilde{Z}} / |\ddot{\tilde{Z}}| \\ \tilde{B} &= \tilde{T} \times \tilde{N}\end{aligned}\quad (12)$$

where the dot indicates differentiation with respect to, ξ , the parametric coordinate. It is also a routine matter to compute surface normals, areas, volumes, curvatures and in short any geometric property.

Table 3 Geometry construction directives *

LARPC	PATCHB	HPB	SCALP
LINEB	PATCHGR	HPL	SCALPH
LINECS	PATCHL	HPHEX	TMOVE
LINEPC	PATCHO	HPN	
	PATCHQ	HPP	
	PATCHR	HPR	
		HP2PAT	
		HP6PAT	

* Mnemonic Suffixes:

A = Algebraic , B = Geometric , CS = Cubic Spline, L = Line,
N = Normal , P = Point , PC = Parametric Cubic ,
Q = Quadrilateral , R = Rotation ,

3.2 Finite Element

The decision to use the sixty-four point isoparametric finite element for the present study rests on its modeling efficiency for anisotropic materials, for pathological shapes and for general boundary conditions. There are many cases in which these qualities are not required; however, the focus here is on the anomalous behavior of composites particularly in highly stressed states. In this regard it is important to realize that material anisotropy affects matrix conditioning as strongly as geometric aspect ratio. Consider, for example, a heated disk with isotropic properties ($E/G = 2.6$) and with highly anisotropic properties ($E/G = 17.25$) in which the temperature varies quadratically with radius. This change in material anisotropy

caused the number of elements required for good stresses to double as Table 4 demonstrates. The large hoop stress ratio at $r = 2R/3$ is near a $\sigma_\theta = 0$ point

Table 4 Anistotropy effects on stress accuracy

	<u>Isotropic</u>		<u>Anisotropic</u>		<u>Anisotropic</u>	
	<u>Two Elements</u>		<u>Two Elements</u>		<u>Four Elements</u>	
r/R	σ_r^*	σ_θ^*	σ_r^*	σ_θ^*	σ_r^*	σ_θ^*
0	.998	.998	1.128	1.128	1.033	1.033
1/3	.981	.977	.841	1.142	1.035	1.045
2/3	.977	1.030	.461	.597	.901	1.365
1	**	.998	**	.473	**	1.099

* Ratio of computed stress to exact stress
 ** The exact stress is zero at this point

and is simply an indication of large differences in small numbers. Similar studies on the effect of high aspect ratios and degenerate element shapes indicate the sixty-four point element has a broad band over which it can maintain good stress accuracy. In order to realize these benefits many practical problems associated with large element matrices had to be solved in the original PATCHES-III system and the added burden of material nonlinearity required similar efforts.

An examination of the relative costs of generating an element stiffness matrix vs. generating an element thermal load vector for a carbon-carbon material range from 40:1 to over 90:1 depending on the geometry. These ratios are very high in comparison to lower order elements and strongly suggest the use of a pseudo-force formulation for material nonlinearity. When direct matrix solution methods are used, another factor in favor of this approach is the ratio of matrix assembly and decomposition time to the forward-backward solution time which is typically 6:1. The computational factors in favor of an initial stress formulation are faster convergence and greater stability at high strain levels. Studies of the relative convergence rates by Havner²⁶ indicate roughly 2:1 ratios in favor of the initial stress method which is not nearly enough to make the method competitive. Stability will not be a factor for small strains and a periodic initial stress cycle can always be taken if necessary.

There are many derivations of the matrix equations associated with the pseudo-force method and they need not be repeated here. The matrix equations used in the present study are very similar to those used by Havner.²⁶

$$\begin{aligned} [K(V, T)] \tilde{U} &= \tilde{F} \\ [K(O, T)] \tilde{U} &= \tilde{F} + [K(O, T) - K(V, T)] \tilde{U} \end{aligned} \quad (13)$$

which leads to the recursion relations

$$\begin{aligned} [K(O, T)] \tilde{U}_{n+1} &= \tilde{F} + [K(O, T) - K(V_n, T)] \tilde{U}_n \\ [K(O, T)] \tilde{U}_{n+1} &= \tilde{F} + \tilde{Q}_n \end{aligned} \quad (14)$$

The pseudo-force term is obtained by integrating the strains from cycle n against the difference in properties referring to Figure 5

$$\tilde{Q}_n = \int_V [B]^T [\Delta C_n] \epsilon_n dV \quad (15)$$

where $[B]$ transforms mesh point displacements \tilde{U} into strains.

A sufficient condition for convergence of the method inferred from Havner's analysis is that $[K(V, T)]$ must be monotonic in V . It should be noted that in material models with inelastic volume changes, the body forces for thermal stress problems will also change. These change can be accounted for in Q_n by using the mechanical strains in Equation (15). The limited experience to date with the method has enjoyed rapid convergence. As more complex problems are attempted, it may be necessary to utilize more efficient recursion relations. Another method that was considered in the study casts the problem in the form of a first order differential equation using the residual vector formed from Equation (14).

$$\underline{\tilde{R}} = -[K]\underline{\tilde{U}} + \underline{\tilde{F}} + \underline{\tilde{Q}} \quad (16)$$

$$\dot{\underline{\tilde{R}}} + \omega \underline{\tilde{R}} = \underline{\tilde{Q}} \quad (17)$$

The dot indicates a time derivative where time increments are synonymous with load increments and the scalar ω is equivalent to an over-relaxation factor. Stricklin, et al.²⁷ use this approach to derive a self-correcting procedure that worked very well in their nonlinear material applications.

3.3 Matrix Solution

The third major computational problem after data generation and matrix generation is matrix solution. The use of a 64-point finite element leads to matrix equations that are relatively dense. It is not unusual for densities of 35 percent to occur in matrices well over dimension 1000. The original linear code development concentrated on the first two computational problems and used existing (NASTRAN) matrix routines to solve the matrix equations. As a result, the system is efficient for problems up to about dimension 1000 and loses efficiency for larger matrices until eventually saturating the computer. The need to solve similar dimension nonlinear matrix equations led to a change to an iterative solution method. The advantages of this change are 1) elimination of large matrix files, 2) freedom from connectivity optimization problems, 3) relatively small core requirements for large problems and 4) heavy vector processing. The disadvantages are possible slow convergence and, in linear problems, N multiple-load conditions are N times as expensive as one. The iterative method selected is the conjugate gradient algorithm²⁸

and in properly scaled coordinates, it converges in far fewer cycles than the dimension of the matrix. Referring to Equation (16) for the residual \tilde{R}

$$\begin{aligned}\tilde{P}_0 &= \tilde{R}(\tilde{U}_0) \\ \tilde{U}_{i+1} &= \tilde{U}_i + t_i \tilde{P}_i \\ \tilde{P}_{i+1} &= \tilde{R}(\tilde{U}_{i+1}) + \beta_i \tilde{P}_i\end{aligned}\tag{18}$$

where the scalars t_i and β_i are

$$\begin{aligned}t_i &= |\tilde{R}_i|^2 / \tilde{P}_i^T [K] \tilde{P}_i \\ \beta_i &= |\tilde{R}_{i+1}|^2 / |\tilde{R}_i|^2\end{aligned}\tag{19}$$

This is the linear form of the algorithm appropriate to the solution of Equation (14). It is also possible to use the nonlinear version of the algorithm to solve Equation (13) directly. In this case, t_i is the smallest positive root of $\tilde{P}_i^T \tilde{R}(\tilde{U}_i + t \tilde{P}_i)$; however, past experience favors successive elastic solutions and this was done. Recent applications of the linear algorithm to 3D composite material problems by Dana²⁹ always converged to four places in less than $N/5$ cycles for problems of dimension $N = 500$ to over 2000. Applications in the present study have been to small one and two element models requiring $N/2$ cycles for similar accuracy.

Efficiency is also a function of the cost per cycle and this is where the vector processing efficiency of scientific computers helps iterative algorithms. Even the CDC 6600 can be made to compute dot products very efficiently by taking advantage of multiple arithmetic units. Unfortunately, older computers like the UNIVAC 1108 do not have this feature and because of their short 32 bit word, all arithmetic must be done in double precision. Several attempts to use mixed mode arithmetic on a UNIVAC 1108 were unsuccessful. The operation most sensitive to round-off error is the transformation of an element stiffness matrix from geometric format to point format. The reason is the large difference in the magnitude of displacements and displacement derivatives. Fortunately this can be avoided by transforming directly from algebraic format to point format. Another

factor affecting cycle efficiency is the amount of data transfer. In the present approach, only the reduced element matrices are transferred into core each cycle. Older versions of the program transferred the reduced structural matrix into core each cycle and this was more expensive primarily because the matrix had to be unpacked a few columns at a time. As a fringe benefit of using element matrices, the data transfer per cycle grows linearly with the number of elements in the model. The overall cycle cost then grows at most linearly since vector processing also increases at most linearly with the number of elements.

A constant issue in iterative methods is the convergence criteria. The first parameter to converge is the energy with the maximum modulus displacement component usually a close second. When these two parameters have converged to seven places, the stresses and strains have about three-place accuracy. Several additional cycles are required to produce stresses and strains that agree with the direct solution to six places (see Figure 6) and this raises the issue. Should these additional cycles be executed to increase the accuracy of the least significant stresses and strains and provide consistency with direct solutions? In linear problems this is done because the increase in cost is modest. However, in nonlinear problems, this expense is harder to justify since the intermediate solutions have no function other than to provide accurate state variables, like strain energy, so that accurate material properties can be computed.

4.0 APPLICATIONS

4.1 Carbon-Carbon Unit Cell

The processing of 3D carbon-carbon materials ideally results in an orthogonal array of fibers in which the open regions are filled with a carbon matrix material during densification. The repetitive volume element for the material has three planes of symmetry and Figure 7 shows schematically one octant of the so-called unit cell. Macroscopic properties for this material are usually based on an analysis of the unit cell with uniform traction or displacement boundary conditions as described by Ross³⁰. A constant property finite element model of this composite would require a minimum of eight elements and the referenced analysis used twenty-seven elements. This same unit cell was modeled with only one variable property element using the techniques described earlier. The constituent properties for the fiber bundles and matrix material are given in Table 5 and a more complete description of the composite is given by Ross³⁰. The results from the one element model are in remarkably good agreement with the results from other analyses presented in Table 6. There are several comments that need to be made about these results to maintain perspective: (1) the orthogonal nature of the weave allows the macroscopic properties to be computed from coarse models (2) the computer plot, Figure 8, of the free thermal expansion of the one element PATCHES-III model shows mesh lines at $\xi = L/6$ to enhance viewing and (3) imperfections in the unit cell can result in large differences between ideal and real material properties.

Given this perspective, one conclusion to be drawn is that without variable property modeling an analysis using the present element would cost over an order of magnitude more and not substantially change the mechanical properties. A second interesting conclusion is related to the poor estimate of α_{11} and the sensitivity of the macroscopic properties to constituent properties. The parametric cubic property modeling used $P(0) = P_0$, $P(1) = \dot{P}$, and $\dot{P}(0) = P(1) = 0$ boundary conditions which tend to lower the high modulus constituent data and raise the low modulus data. A sensitivity study revealed that α_{11} was primarily a function of the longitudinal

Table 5 Unit cell dimensions and constituent properties

L12 = 0.7, L13 = 0.3, L21 = 0.035, L23 = 0.015, L31 = L32 = 0.0062

	Z_1^*	Z_2^*, Z_3^*	Matrix
$E_L \times 10^{-6}$	33.34	4.841	0.70
$E_T \times 10^{-6}$.8958	1.332	0.70
ν_{LT}	.1818	.1840	0.15
ν_{TT}	.1898	.1849	0.15
$G_{LT} \times 10^{-6}$	1.169	.9405	0.30435
$G_{TT} \times 10^{-6}$.3765	.5619	0.30435
$\alpha_L \times 10^6$.5133	1.647	2.5
$\alpha_T \times 10^6$.5149	4.571	2.5

* Z_i Fiber bundle properties, reference 30.

Table 6 Unit cell macroscopic property comparisons

	PATCHES-III (1-Element)	SAP* (27-Elements)	RULE-OF- MIXTURES
$C11 \times 10^{-6}$	11.793	12.461	12.460
$C12 \times 10^{-6}$.274	.226	.230
$C13 \times 10^{-6}$.248	.214	.222
$C22 \times 10^{-6}$	2.274	2.328	2.340
$C23 \times 10^{-6}$.251	.211	.225
$C33 \times 10^{-6}$	1.518	1.376	1.416
$C44 \times 10^{-6}$.846	-----	.853
$C55 \times 10^{-6}$.760	-----	.754
$C66 \times 10^{-6}$.625	-----	.609
$\alpha 11 \times 10^6$	1.462	.748	.755
$\alpha 22 \times 10^6$	2.896	2.567	2.271
$\alpha 33 \times 10^6$	4.284	3.924	4.039

*SAP results from reference 30.

α_L of the Z_1 fiber which is very small. In this case the parametric cubic model effectively doubled the α_L of the Z_1 and the boundary conditions for this property should be adjusted to give the correct area under the curve.

In addition to featuring the intra-element modeling capability, this application also raises the issue of sensitivity analyses. Schmit³¹ has long advocated making analysis programs more design oriented by providing information on the sensitivity of the analysis results to design changes. The present formulation of the nonlinear material effects, Equation (16), allows such information to be computed by making the change in material properties constant with respect to a state variable and halting the iteration after the first or second cycle. This procedure was successfully applied to a unit cell analysis as part of another study.

4.2 Cracked Bar

To establish the utility of the parametrization in Table 3 for three-dimensional crack analyses, an internally cracked bar, Figure 9, studied by Gyekenyesi and Mendelson³² was analyzed using a two element symmetry model. This problem is a finite dimensional version of the penny shaped or poker chip crack problem whose elasticity solution contains a square root strain singularity. The two element PATCHES-III model shown in Figure 10 uses parametrizations in the axial and radial directions that induce a square root strain singularity. When the singularity is at $Z(1)$ rather than $Z(0)$, as in the element over the crack face, the formal expression for $Z(\xi)$ is

$$Z(\xi) = a(2\xi - \xi^2) \quad (21)$$

which lead to the same behavior at $Z(1)$

$$\begin{aligned} u_{,r} &= u_{,\xi} \xi_{,z} Z_{,r} \\ &= (a/2\sqrt{r}) u_{,\xi} \end{aligned} \quad (22)$$

where $r = 1 - Z_1$ in this element and the comma notation indicates differentiation.

There are two features of the parametric cubic model that merit attention 1.)

the geometry model is trivial to construct in algebraic or geometric format and

2.) the element displacement functions in geometric format have $u_{,\xi}(1)$ as a nodal variable which is directly related to the stress intensity factor. The coefficients for the algebraic, S_i , and geometric, B_i , representations of Equation (21) are simply

$$S_1 = 0, S_2 = -a, S_3 = 2a, S_4 = 0 \quad (23)$$

and

$$B_1 = 0, B_2 = a, B_3 = 2a, B_4 = 0 \quad (24)$$

The construction of the complete geometry model for this problem required six grid cards, two PATCHB directives containing only four nonzero B_{ij} each, and two HPR directives. However, this easily constructed model is capable of accurately representing the highly deformed geometry, Figure 11, as the crack opens under an axial load.

Comparison between the present solution and the Gyekenyesi-Mendelson solution, Figure 12, show good stress agreement. The strains in PATCHES-III are computed at the Gaussian points and then transformed to the one-third points which accounts for the finite amplitude stresses at $r = 0$. The stress intensity factor computed from $u_{,\xi}(1)$ of the element containing the crack face is 2.5% higher than that for an infinite dimension bar which is slightly closer than the Gyekenyesi-Mendelson result. The displacements are also correspondingly closer to classical than theirs. The parametric derivative $u_{,\xi}$ in the present model may provide a convenient characterization of stress intensity. The issue would depend on its behavior under admissible reparametrizations such as those in Table 2. Since in general the strength of the singularity is unknown and changes as a function of material anisotropy³³. A carpet plot of the radial strain component over the (Z_1, Z_3) plane, Figure 13, shows a theta dependence in keeping with the elasticity solution which contains $\sin \theta$ and $\cos \theta$ terms in the strains.

4.3 Graphite Bar

Consider next the application of a stress-strain state variable model to a composite material whose inelastic behavior is known to differ significantly from that

of metals. ATJ-S graphite is one such material and its inelastic behavior has been extensively tested by Jortner² in biaxial and triaxial stress states. To focus clearly on the ability of the model to represent inelastic behavior not normally found in metals, a case of hydrostatic compression was selected. A classical plasticity solution would be identical to the elasticity solution for hydrostatic loading. This material, however, is quite nonlinear under hydrostatic compression and the Batdorf model for ATJ-S fits the Jortner test data extremely well, Figure 14. A transversely isotropic graphite bar, diameter 0.250 inches by 4.0 inches was loaded by an axial force and external pressure in these tests. The stress-strain version of the Batdorf model was used to compute the data in Figure 14 assuming constant strain ratios which is consistent with the Jortner results. The bar elastic constants and F_{ijkl} for the Batdorf model are given in Table 7 where the across grain direction coincides with the centerline of the bar and the σ^0 parameters are $\sigma_r^0 = \sigma_\theta^0$ and σ_z^0 in,

$$V_2 = C^0 \left\{ \left[\left(\frac{\sigma_z}{\sigma_z^0} \right)^2 + \left(\frac{\sigma_\theta}{\sigma_\theta^0} \right)^2 + \left(\frac{\sigma_r}{\sigma_r^0} \right)^2 + 2 \left(\frac{\sigma_z \sigma_\theta}{\sigma_z^0 \sigma_\theta^0} + \frac{\sigma_z \sigma_r}{\sigma_z^0 \sigma_r^0} + \frac{\sigma_r \sigma_\theta}{\sigma_r^0 \sigma_\theta^0} \right) \right]^{1/2} - 1 \right\} \quad (25)$$

These constants are determined from uniaxial data including the constant multiplier, C^0 , where $C^0 = 0.15$ for ATJ-S. The use of constant strain ratios in the axial tension range causes the stress ratios from the Batdorf model to deviate slightly from ± 1 but this difference is small even up to $\epsilon_z = 0.004$ as Figure 14 demonstrates.

Table 7 Graphite bar material constants

DIRECTION	E	ν	G	C^0
With Grain (r, θ)	1.8×10^6	0.1	0.8×10^6	1.22×10^3
Across Grain (z)	1.2×10^6	0.1	0.65×10^6	1.04×10^3

A PATCHES-III model of the gage section of the test specimen was developed using the material constants from Table 7. The geometric simplicity and transverse isotropy allowed a simple one element symmetry model of a 30° segment of the bar to be used. A single hydrostatic load condition of 4 ksi compression was analyzed using successive elastic solutions of Equation (15) starting from the linear solution. Convergence of the pseudo-force method was rapid as Figure 15 illustrates and the resulting strains are in good agreement, Figures 16 and 17, with Jortner's test data. In this particular problem the exact stress solution is known from equilibrium considerations and the iteration was terminated when the stresses were within one percent of the exact solution. The convergence characteristics of several interesting parameters are shown in Table 8.

Table 8 Nonlinear solution convergence data

Cyle	σ	Potential Energy	V_2	Q_{\max}
0	-4000	-.00614	.7537	0.0
1	-3748	-.00973	.5189	2.33
2	-3845	-.00119	.6335	3.52
3	-3909	-.00132	.6882	4.17
4	-3948	-.00139	.7171	4.54
5	-3970	-.00143	.7331	4.75
∞	-4000	-----	.7337	-----

The computational performance of the model on an UNIVAC 1108 was good, taking less than one minute per pseudo-load cycle for this small problem and should be better on a parallel processing machine. The number of conjugate gradient cycles required to maintain constant solution accuracy decreased with each pseudo-load cycle but not dramatically. The original linear solution starting from zero used 74 cycles and the last solution required 54 cycles. The time required to generate Q_n each cycle was negligible in comparison to the matrix solution costs.

The computational system designed and developed in this study has the flexibility to encompass most of the available inelastic composite material models and appears to have the efficiency to make its use feasible if not practical. It will require additional experience and testing in a variety of applications before the efficiency of the system can be fully established but the basic design philosophy seems sound; namely, better computers are a more likely development than better numerical methods. The use of stress or strain state variable modeling is certainly not new but the design of one standard input format for a entire class of material models is a step forward.

The introduction of isoparametric modeling for material properties as well as geometry is another important step required by the variability of composite materials. In each instance mentioned, the results presented are a beginning with much additional work required before the triaxial inelastic behavior of composites is encompassed by the available models. This work should include:

1. Multiaxial testing programs, such as Jortners, for other three-dimensional composites that includes a parallel computational model development effort.
2. Extensions to include interstitial slip with friction at bimaterial interfaces. This behavior in fiber reinforced composites is important to an understanding of microstructural effects on the failure of these materials.
3. Development of alternatives to point stresses and strains as a measure of composite material response for structural applications.
4. Development of representative volume element models suitable for inelastic behavior and failure that account for the statistical nature of three-dimensional composites.

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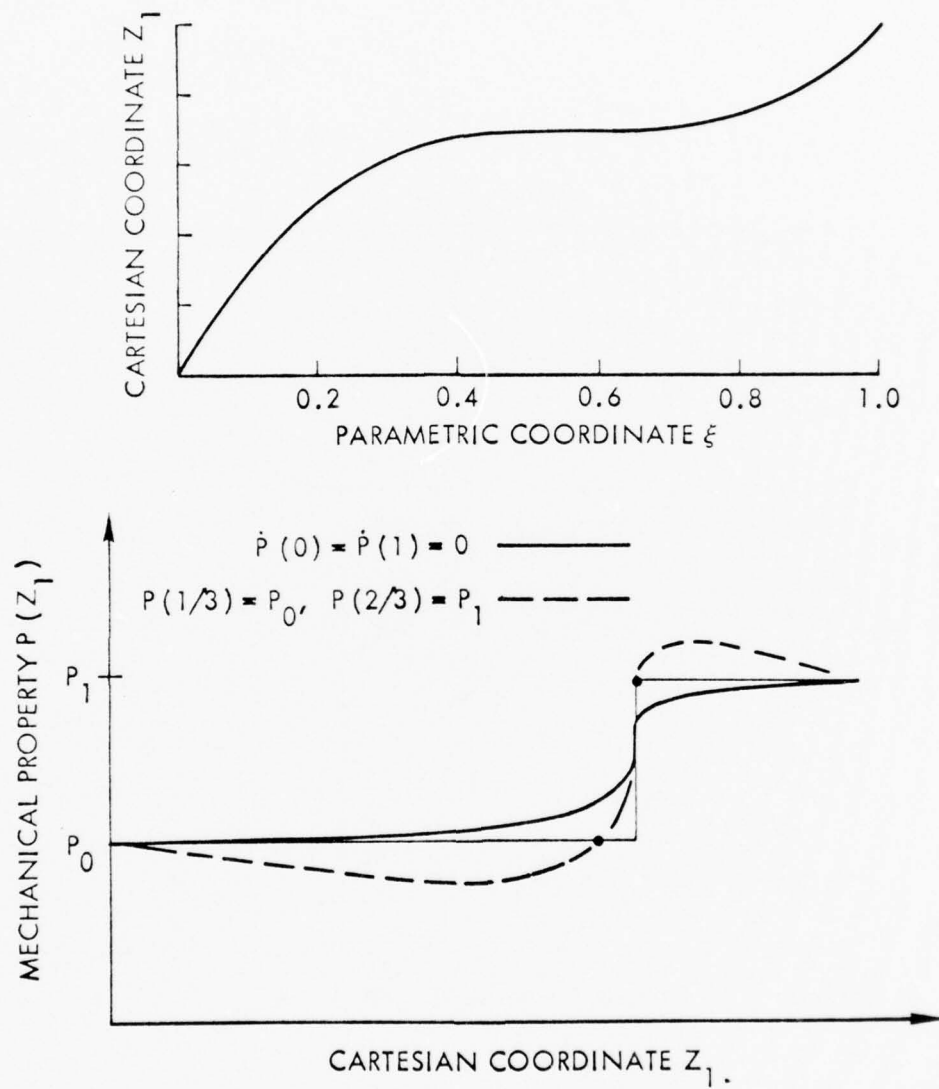


Figure 1. Parametric Cubic Modeling of a Bimaterial Interface

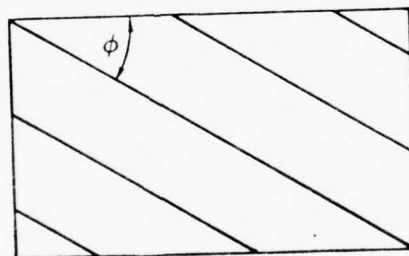
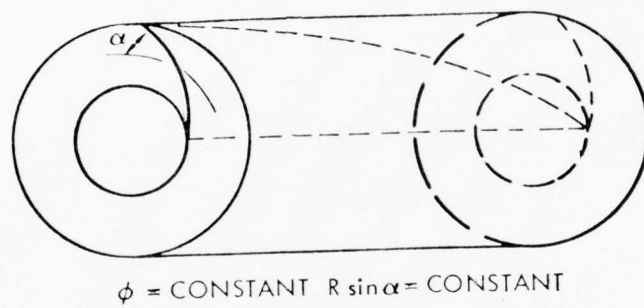


Figure 2. Rosette Construction Materials

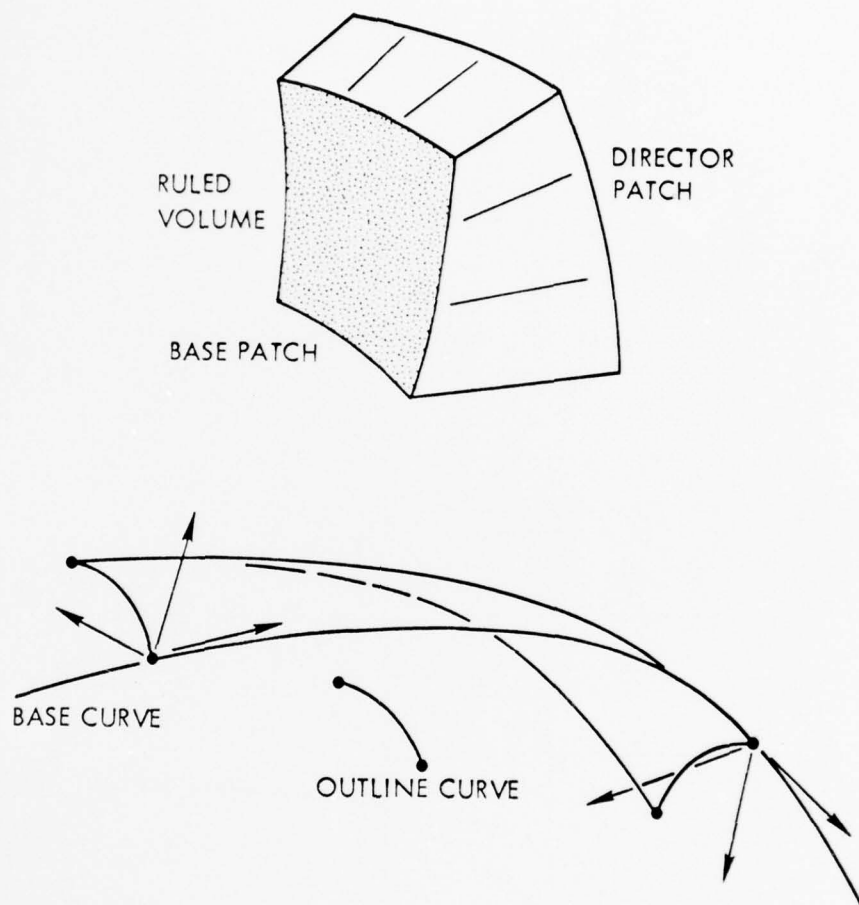


Figure 3. Construction of Finite Geometry Models

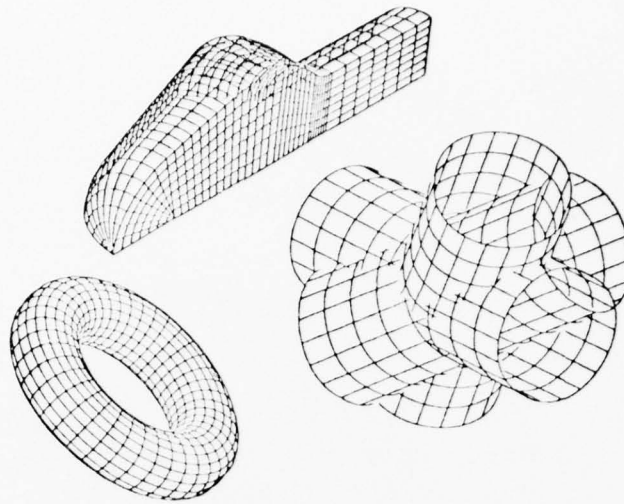


Figure 4. Parametric Cubic Models

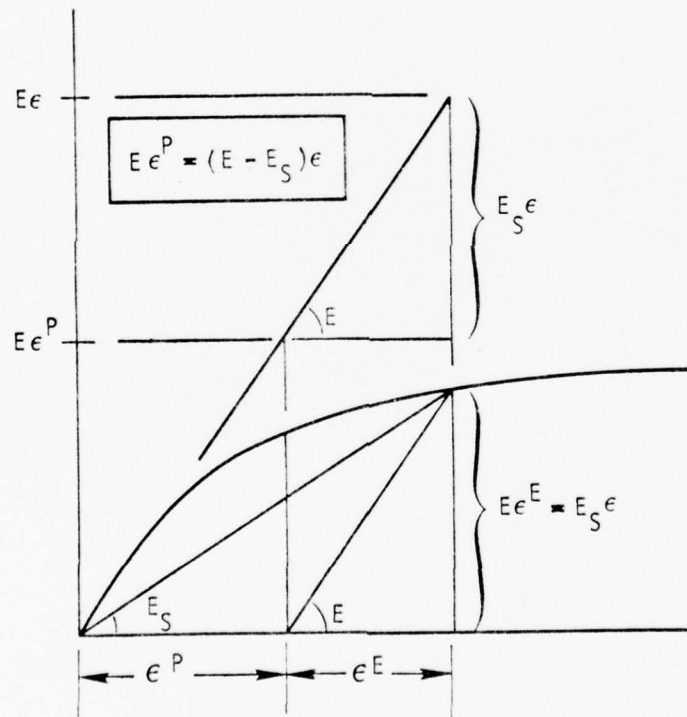


Figure 5. Stress-strain Schematic

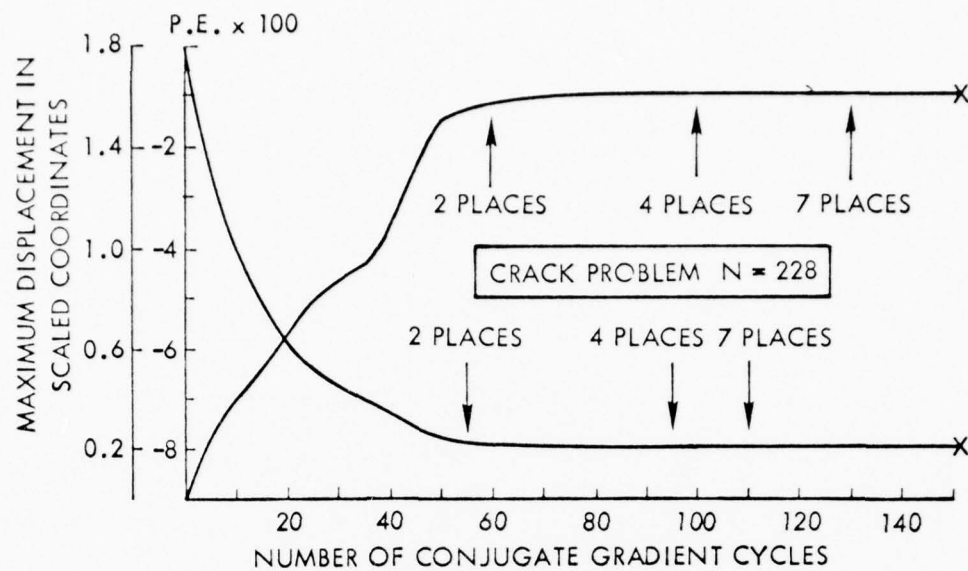


Figure 6. Energy Convergence of the Conjugate Gradient Solution

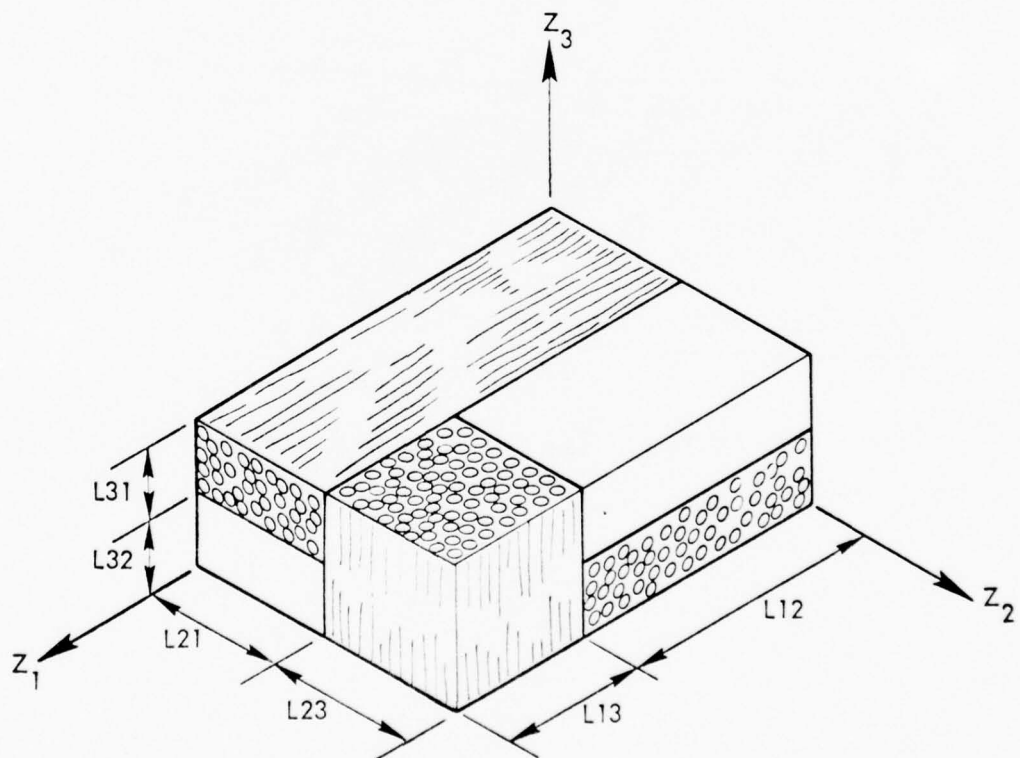


Figure 7. Carbon-carbon Unit cell Schematic

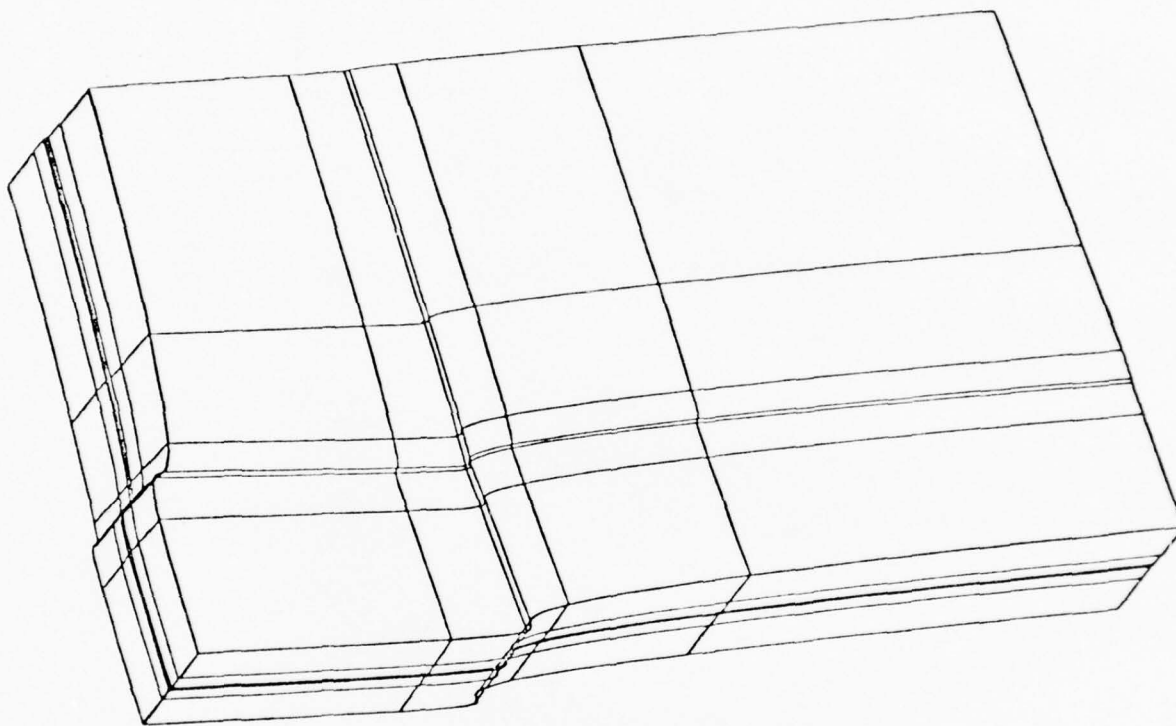


Figure 8. Unit Cell Thermal Expansions

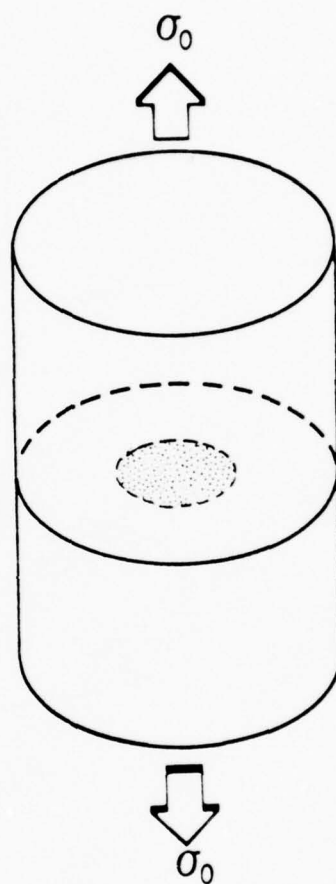


Figure 9. Cracked Bar Schematic

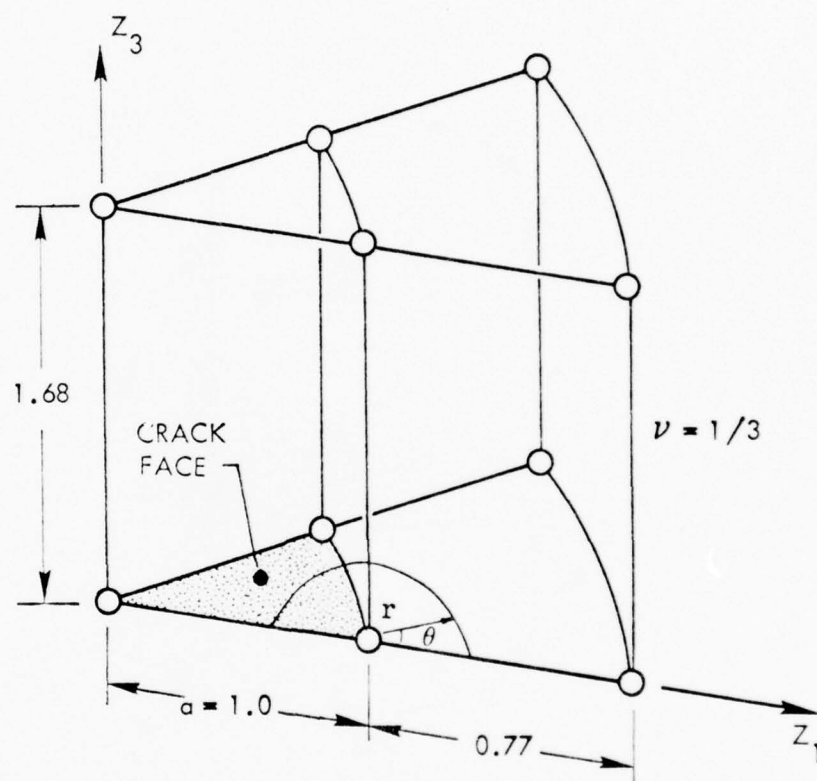


Figure 10. PATCHES-III Model of Cracked Bar

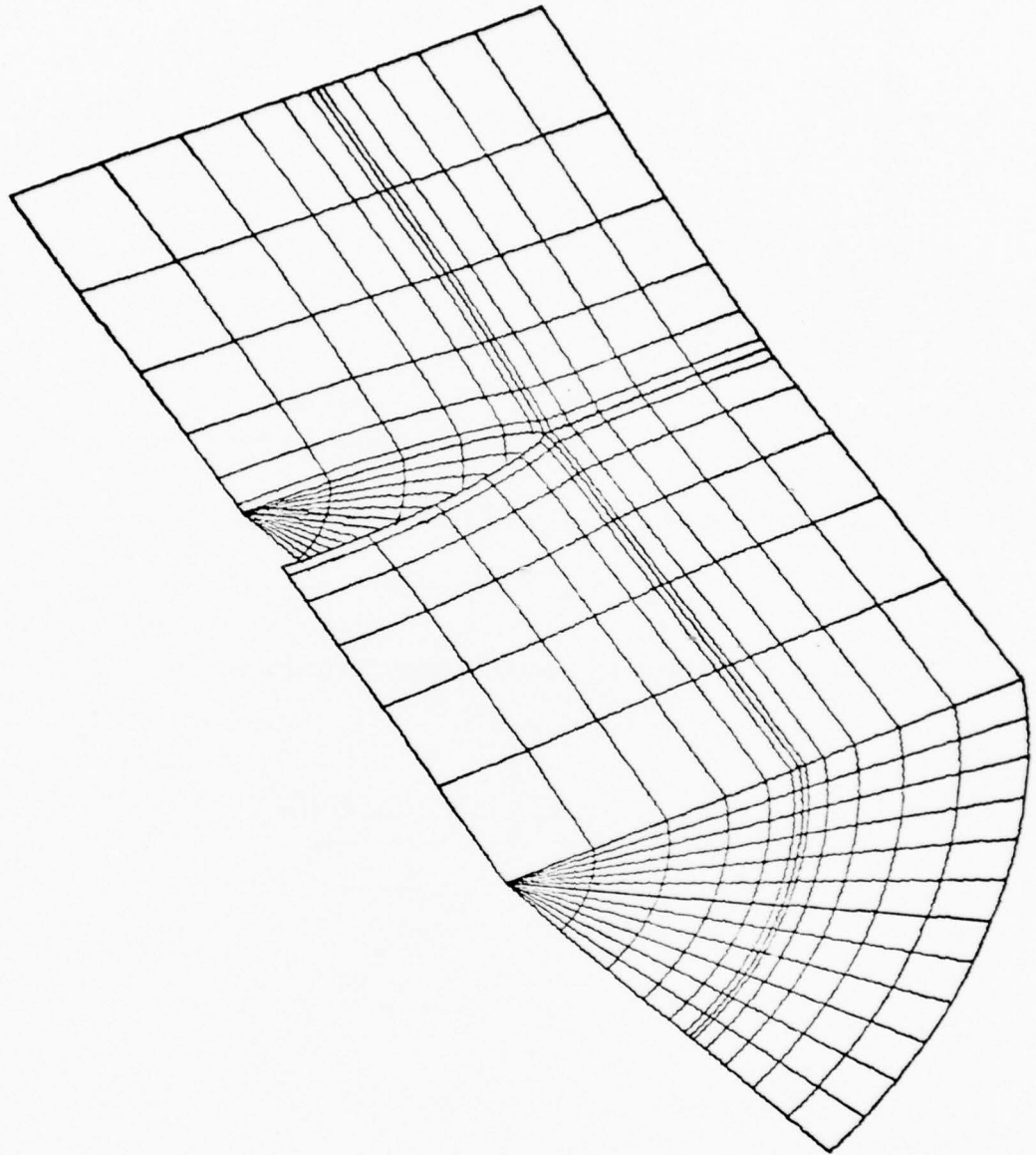


Figure 11. Cracked Bar Elastic Deformations

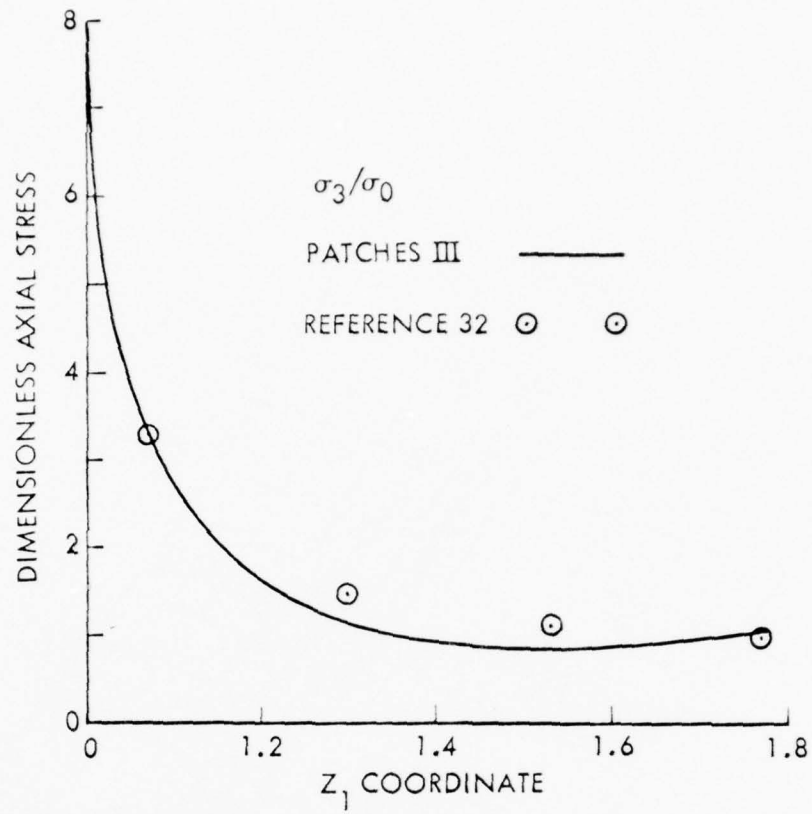


Figure 12. Cracked Bar Axial Stress Comparisons

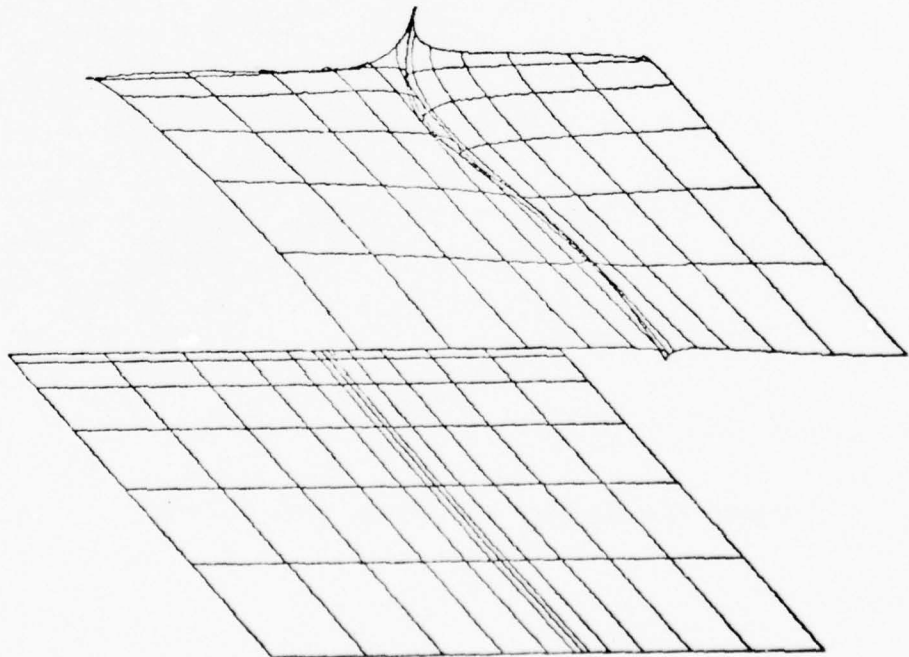


Figure 13. Cracked Bar Radial Strain Display

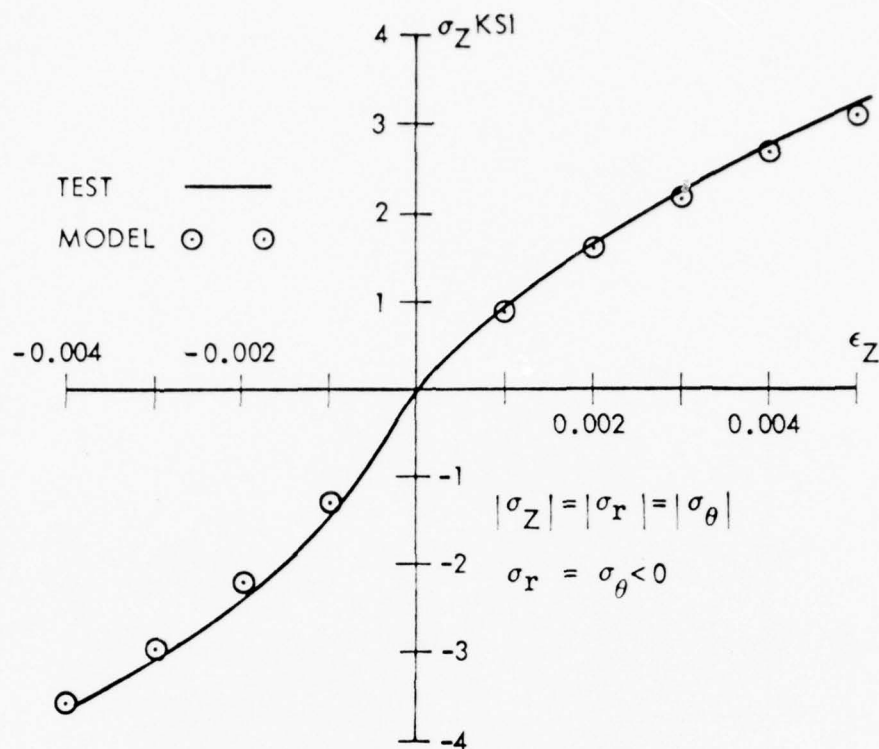


Figure 14. ATJ-S Triaxial Stress-Strain Comparisons

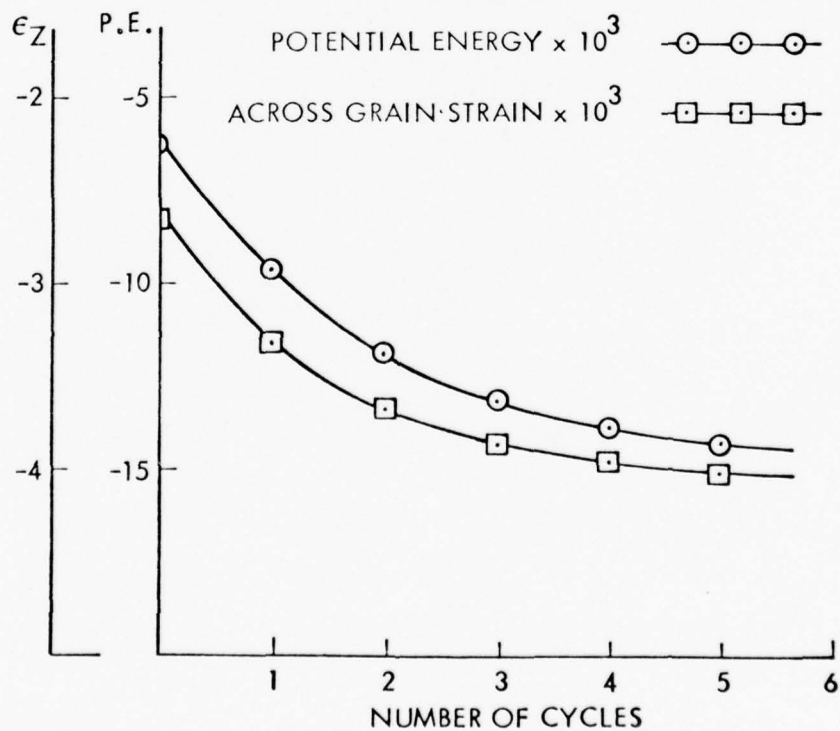


Figure 15. Convergence of Pseudo-force Method

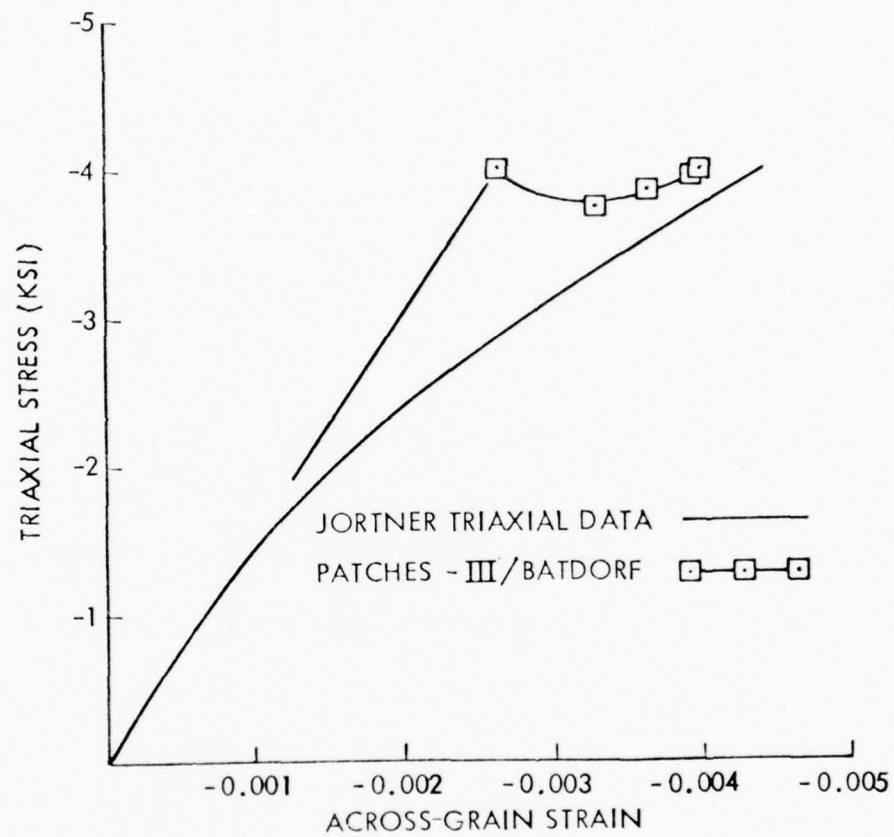


Figure 16. Inelastic Across-grain Strain Convergence

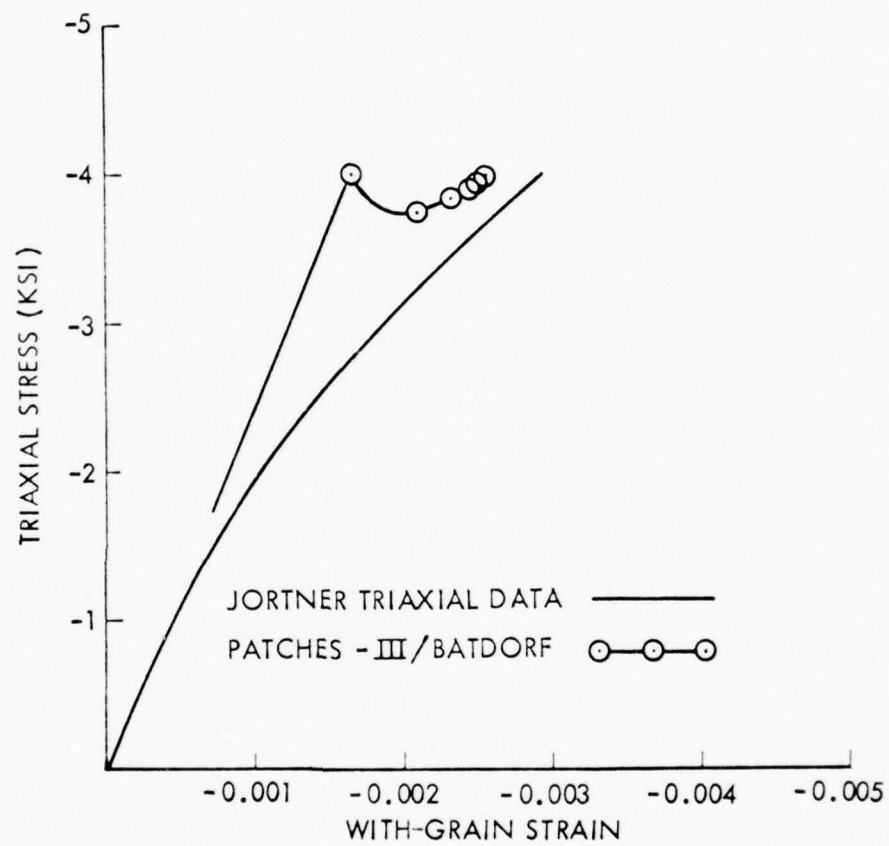


Figure 17. Inelastic With-grain Strain Convergence

APPENDIX

The program updates for nonlinear material modeling in PATCHES-III were made using the generalized postprocessor system GPOSTP. This system interfaces with PATCHES-III through three files: PPDATA, INPT and RANDOM16 created during a normal execution of a linear elastic analysis. Any routine in the PATCHES-III library is available to GPOSTP and allows new capability, like MATN, to be fully developed and tested before restructuring the original code.

In the present effort a major change had to be made to the code to cope with large nonlinear matrix problems. These changes are complete and the restructured code is shown in the update to Figure 4-2 of the programmers manual. The GPOSTP system for the MATN postprocessor is shown in Figure 4-2a. The limited testing of MATN accomplished during the study has been very successful. This link will be added to the basic system after all options are fully tested.

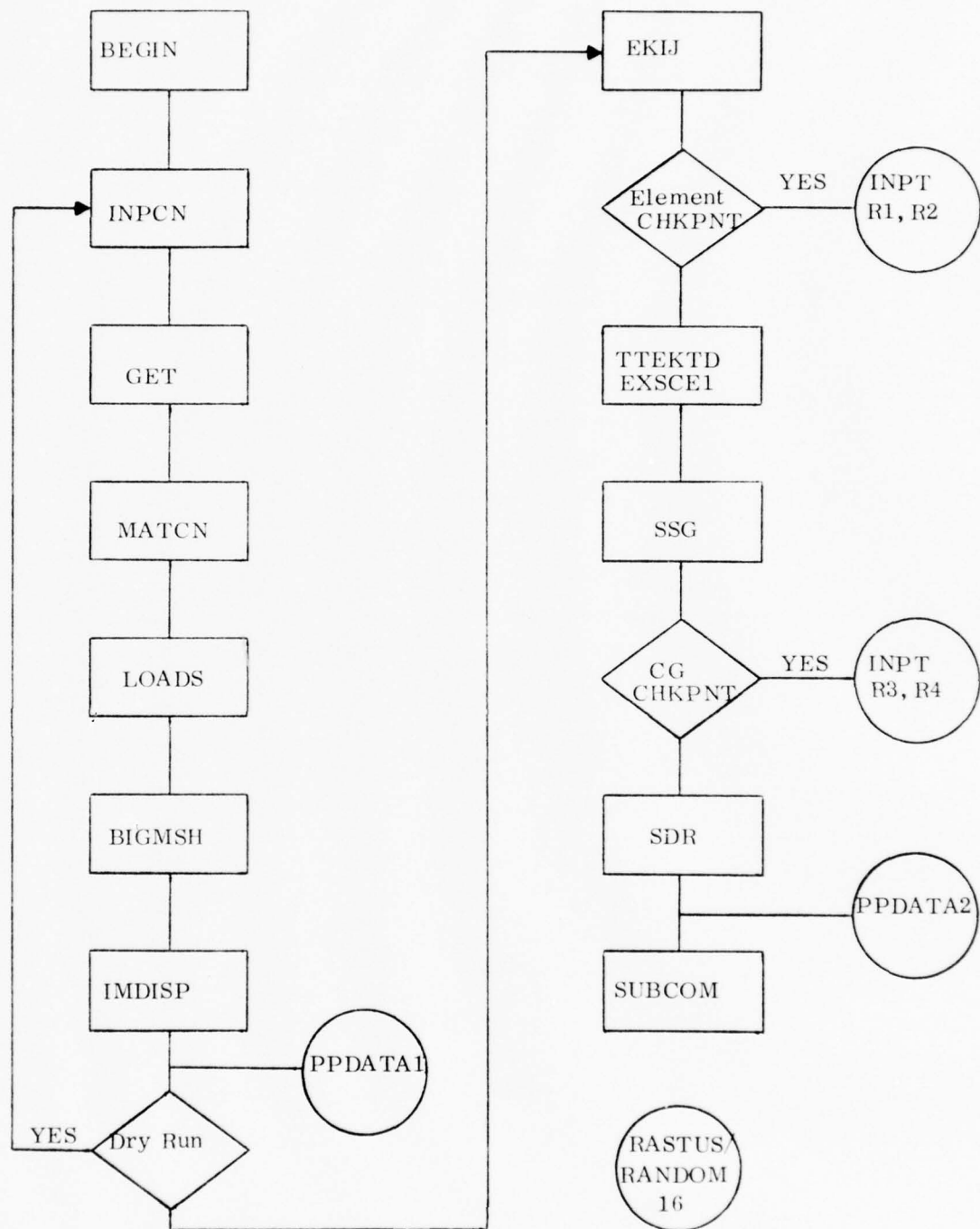


Figure 4-2. Basic Flow PATCHES-III Version 7

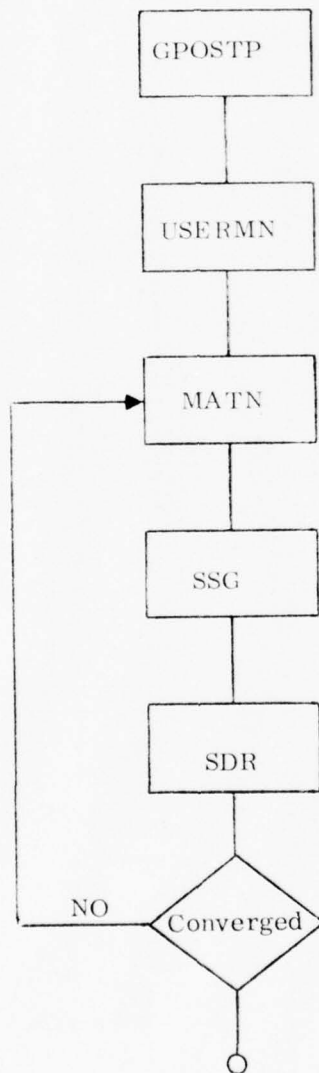
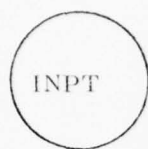


Figure 4-2a. Basic Flow MATN Postprocessor

BULK DATA DECK

Input Data Card: PATCHGR Patch generated by general line rotation.

Description: Generates a bicubic patch for the surface created by rotating a PC line about a general axis of rotation through gamma degrees.

Format and Example:

1	2	3	4	5	6	7	8	9	10
PATCHGR	ID	LID, SEG	ZA1	ZA2	ZA3	ZB1	ZB2	ZB3	+P1
PATCHGR	5	3	1.5	0.	-3.0		2.5	0.3	+P1
+P1	TID	GAMMA	GAMMAO						
+P1		25							

Field

Contents

ID	The identification number to be given the patch generated from line LID, segment number SEG.
LID, SEG	The line number, LID, and segment number, SEG, that identifies the PC line to be rotated. A blank SEG defaults to one.
ZA1, ZB1	Coordinates of two points that define the rotation axis directed from Z_A to Z_B .
TID	Transformation ID, if any, that defines a geometric transformation to be applied to the PC line before rotation. The line, LID, does not change.
GAMMA,	The angle in degrees through which the PC line is rotated starting γ_0 degrees from the initial position of the line. The sense of rotation is determined by the right-hand rule and the directed line (vector) from Z_A to Z_B .

BULK DATA DECK

Input Data Card: PATCHO Outline patch(es)

Description: The patch(es) generated by moving an outline curve along a base curve with a fixed orientation of the outline curve in the global frame or in the local Frenet frame of the base.

Format and Example:

1	2	3	4	5	6	7	8	9	10
PATCHO	ID1	BLID, SEG	OLID, SEG	TID	FRAME	ID2	...	IDN	
PATCHO	5	6, 1	3		F	11	THRU	15	

Field

Contents

- ID1 Patch identification number for first patch generated.
- BLID, SEG Baseline identification number, and segment number, SEG.
If the SEG is not specified, the entire line BLID is used.
- OLID, SEG Outline curve identification number, and segment number, SEG.
If the SEG is not specified, the entire line OLID is used.
- TID Transformation ID, if any, that defines a rotation matrix to reorient the outline curve relative to the base curve. The outline curve is always translated to the first grid point of the *baseline independent of TID*.
- FRAME E for fixed outline orientation with respect to the Cartesian frame.
F for fixed outline orientation with respect to the local Frenet frame of the base curve.
- ID2, 3, ..., N List of identification numbers to be given the second and subsequent patches generated, if any. This sequence proceeds from the second line segment of OLID to last and then repeats from the first segment for the next segment of BLID.

BULK DATA DECK

Input Data Card: MATOR Orthotropic Material Definition

Description: Defines the material properties for a linear, temperature independent, orthotropic material from engineering constants

Format and Example:

1	2	3	4	5	6	7	8	9	10
MATOR	MID	FRAME	POINTS	O1	O2	..	ON		
MATOR	3	1	8	2	6	THRU	12		

<u>Field</u>	<u>Contents</u>
MID	Material identification number
FRAME	=1, Properties are in an orthonormal Cartesian frame =2, Properties are in the normalized parametric frame of the element. Assumes parametric frame is quasi cylindrical or speherical.
POINTS	=1, Constant material properties (N=1). =8, Trilinear variation of material properties (N=8). =64, Tricubic variation of material properties (N=1).
O1,O2, ..., ON	Matrix identification number for the orthotropic material engineering constants at the interpolation points. If POINTS is equal 64 a single entry is used to identify a matrix containing the 64 OID's.
Remarks:	<ol style="list-style-type: none"> The engineering constants are entered in sequence E_{11}, E_{22}, E_{33}, ν_{12}, ν_{13}, ν_{23}, G_{12}, G_{13}, G_{23} on the MTRX matrix card (s). The Air Force Design Guide convention for Poisson ratios is used i.e. $E_{ii} \nu_{ji} = E_{jj} \nu_{ij}$.

BULK DATA DECK

Input Data Card: MATN Nonlinear Material Properties.

Description: Defines the nonlinear behavior of a material property in terms of stress or strain state variables.

Format and Example:

1	2	3	4	5	6	7	8	9	10
MATN	MID	MPID	MAT	SHAPE	STATE1	STATE2			+M1
MATN	10		MAT0	CC	V2				+M1
+M1	TI	MTRX-M	V1	MTRX-V1	V2	MTRX-V2			+M2
+M1	500.	6	100.0	6					+M3

Field

Contents

MID	Material identification number.
MPID	Identification number of the matrix that defines the mesh points which have these properties. The default is all mesh points in an element.
MAT	Mnemonic that defines the format of the data in MTRX-M.
SHAPE	Mnemonic that defines the interpolation method for T, V1, V2 in that order; CLL, LCC, etc. where L = Linear and C = Cubic.
STATE1	Mnemonic that identifies state variable I (see remarks).
VI	Value of the variable I for which these data apply. A blank field requires MTRX-VI data to define VI.
TI	Temperature at which these data apply. The default is all temperatures.
MATRX-M	Matrix identification number of properties data in the format defined by MAT.
MTRX-VI	Matrix identification number of coefficients that define VI.
<u>Remarks</u>	<ol style="list-style-type: none"> 1. The standard state variables are VI and VBI for $I = 1, 2, 3, 4$ where $V1 = F_i \epsilon_i$, $VB4 = F_{ijkl} \sigma_i \sigma_j \sigma_k \sigma_l$, etc. 2. The only non-standard options are the Batdorf model identified by a B and the Lee identified by an L.

BULK DATA DECK

Input Data Card: TMOVE Rigid Body Transformation

Description: Defines a transformation that move objects (lines, patches, hyper-patches) as rigid bodies.

Format and Example:

1	2	3	4	5	6	7	8	9	10
TMOVE	ID	Z01	Z02	Z03	DCID	Ψ	θ	ϕ	+TA
TMOVE	1	3.0	0.	3.		30.	20.	-10.	+TA
+TA	T1	T2	T3						
+TA	3.0	0.	4.0						

Field

Contents

ID Transformation identification number (1 to 100).

Z01 Defines an origin for rotation of the object.

DCID Direction cosine matrix identification number. If blank or zero, the Euler angles Ψ , θ , ϕ , define the rotation matrix.

Ψ, θ, ϕ Euler angles in the 3, 1, 3 rotation sequence.

$$R = \begin{pmatrix} \cos \psi \cos \phi - \cos \theta \sin \phi \sin \psi & -\sin \psi \cos \phi - \cos \theta \sin \phi \cos \psi & \sin \theta \sin \phi \\ \cos \psi \sin \phi + \cos \theta \cos \phi \sin \psi & -\sin \psi \sin \phi + \cos \theta \cos \phi \cos \psi & -\sin \theta \cos \phi \\ \sin \theta \sin \psi & \sin \theta \cos \psi & \cos \theta \end{pmatrix}$$

TI Defines a translation to be applied after the rotation.

Remarks: 1. The complete transformation can be defined as

$$\tilde{Z}^* = R (\tilde{Z} - \tilde{Z}_0) + \tilde{T}$$

PROTOTYPE DEVELOPMENT ASSOCIATES
VERSION 7.2

PROTOTYPE DEVELOPMENT ASSOCIATES
VERSION 7.2

Carbon-Carbon Unit Cell
Bulk Data Model

CARD	R	U	L	K	D	A	T	A	C	A	R	D	S	PAGE 1
1-	SOC10	10	1	3	5	6	7	8						
2-	SOC10	10	1	2	2	3	7	6						
3-	SOC10	10	1	1	4	3	7	3						
4-	SOC10	10	1	1	1	3	7	3						
5-	SOC10	10	1	2	1	4	5	2						
6-	SOC10	10	1	3	1	3	4	3						
7-	SOC10	20	1	1	1	5	6	2						
8-	SOC10	20	1	1	4	3	7	3						
9-	SOC10	20	1	2	1	4	7	3						
10-	SOC10	20	1	2	2	3	7	3						
11-	SOC10	20	1	3	1	2	7	3						
12-	SOC10	20	1	3	1	2	7	3						
13-	SOC10	20	1	3	1	2	7	3						
14-	SOC10	30	1	1	1	5	6	2						
15-	SOC10	30	1	2	1	4	5	2						
16-	SOC10	30	1	2	2	3	7	3						
17-	SOC10	30	1	3	1	2	7	3						
18-	SOC10	30	1	3	1	2	7	3						
19-	SOC1	30	1	7	1	6	1.24=4	8						
20-	MTX-15	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
21-	+X1	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
22-	+X2	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
23-	+X3	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
24-	+X4	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
25-	+X5	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
26-	+X6	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
27-	+X7	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
28-	MTX-16	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
29-	+Y1	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
30-	+Y2	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
31-	+Y3	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
32-	+Y4	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
33-	+Y5	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
34-	+Y6	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
35-	+Y7	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
36-	MTX-17	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
37-	+Z1	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
38-	+Z2	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
39-	+Z3	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
40-	+Z4	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
41-	+Z5	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
42-	+Z6	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
43-	+Z7	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
44-	MTX-18	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
45-	+C1	0.0	0.0	0.0	0.0	0.0	0.0	0.0						
46-	MTX-19	0.0	0.0	0.0	0.0	0.0	0.0	0.0						

R U L K O A T A C A R D S

FIELD=1, FIELD=2, FIELD=3, FIELD=4, FIELD=5, FIELD=6, FIELD=7, FIELD=8, FIELD=9, FIELD=10

CARD

47-	MATOR	1	1	64	20	105	106	107	108+M1
48-	MTRX-20	101	102	103	104	113	114	115	116+M2
49-	M1	109	110	111	112	121	122	123	124+M3
50-	M2	117	118	119	120	129	130	131	132+M4
51-	M3	125	126	127	128	137	138	139	140+M5
52-	M4	133	134	135	136	145	146	147	148+M6
53-	M5	141	142	143	144	153	154	155	156+M7
54-	M6	149	150	151	152	161	162	163	164
55-	M7	157	158	159	160	169	170	171	
56-	MATAL	1	1	64	30	205	206	207	208+M1
57-	MTRX-30	201	202	203	204	213	214	215	216+M2
58-	M1	209	210	211	212	221	222	223	224+M3
59-	M2	217	218	219	220	229	230	231	232+M4
60-	M3	225	226	227	228	237	238	239	240+M5
61-	M4	233	234	235	236	245	246	247	248+M6
62-	M5	241	242	243	244	253	254	255	256+M7
63-	M6	249	250	251	252	261	262	263	264
64-	M7	257	258	259	260	269	270	271	
65-	DPATA	5	1	2	3	4			+D1
66-	M1								+D2
67-	M2								100.
68-	DPATEQ	6	5	5	6	7	8		
69-	DPATEQ	7	5	4	5	7	3		
70-	DPATEQ	8	5	2	3	7	6		
71-	DPATEQ	10	5	6					
72-	TEMP	10	1	10					
73-	MTRX-101	1332+07	4841+07	1332+07	5063-01	1849+00	1840+00	9405+06	5619+06+M10100
74-	M10100	9405+06							
75-	MTRX-102	1168+07	3767+07	1168+07	7639-01	1759+00	1752+00	7756+06	4951+06+M10200
76-	M10200	7756+06							
77-	MTRX-103	8639+06	1774+07	8639+06	1242+00	1590+00	1580+00	4893+06	3712+06+M10300
78-	M10300	4693+06							
79-	MTRX-104	7000+06	7000+06	7000+06	1500+00	1500+00	1500+00	3044+06	3044+06+M10400
80-	M10400	3044+06							
81-	MTRX-105	1332+07	4841+07	1332+07	5063-01	1849+00	1840+00	9405+06	5619+06+M10500
82-	M10500	9405+06							
83-	MTRX-106	1211+07	3010+07	1446+07	7874-01	1692+00	1685+00	7929+06	5379+06+M10600
84-	M10600	8183+06							
85-	MTRX-107	0852+06	1805+07	1659+07	1309+00	1400+00	1397+00	5188+06	4933+06+M10700
86-	M10700	5915+06							
87-	MTRX-108	8639+06	1774+07	1590+00	1242+00	1242+00	1242+00	3712+06	4693+06+M10800
88-	M10800	4693+06							
89-	MTRX-109	1332+07	4841+07	1332+07	5063-01	1849+00	1840+00	9405+06	5619+06+M10900
90-	M10900	9405+06							
91-	MTRX-110	1200+07	3889+07	1963+07	8309-01	1568+00	1561+00	8250+06	6173+06+M11000
92-	M11000	8977+06							

R U L K D A T A C A R D S

CARD

FIELD=1, FIELD=2, FIELD=3, FIELD=4, FIELD=5, FIELD=6, FIELD=7, FIELD=8, FIELD=9, FIELD=10

93- MTRX-111, 1211+07, 2120+07, 3136+07, 1434+00, 1045+00, 1043+00, 6106+06, 7202+06+M1100
94- M1100, 8183+06
95- MTRX-112, 1160+07, 1168+07, 3767+07, 1759+00, 7639-01, 7639-01, 4951+06, 7756+06+M11200
96- M11200, 7756+06
97- MTRX-113, 1332+07, 4841+07, 1332+07, 5063-01, 1849+00, 1840+00, 9405+06, 5619+06+M11300
98- M11300, 9405+06
99- MTRX-114, 1332+07, 3931+07, 2242+07, 8544-01, 1501+00, 1494+00, 8423+06, 6601+06+M11400
100- M11400, 9405+06
101- MTRX-115, 1332+07, 2242+07, 3931+07, 1501+00, 8544-01, 4521-01, 6601+06, 8423+06+M11500
102- M11500, 9405+06
103- MTRX-116, 1332+07, 1332+07, 4841+07, 1849+00, 5063-01, 5063-01, 5619+06, 9405+06+M11600
104- M11600, 9405+06
105- MTRX-117, 9630+07, 3818+07, 1219+07, 8464-01, 1841+00, 1855+00, 9997+06, 7193+06+M11700
106- M11700, 7043+06
107- MTRX-118, 9509+07, 3023+07, 1098+07, 1037+00, 1774+00, 1790+00, 8776+06, 6698+06+M11800
108- M11800, 6721+06
109- MTRX-119, 9284+07, 1546+07, 8721+06, 1392+00, 1649+00, 1668+00, 6507+06, 5780+06+M11900
110- M11900, 4453+06
111- MTRX-120, 9162+07, 7508+06, 7508+06, 1582+00, 1582+00, 1603+00, 5286+06, 5286+06+M12000
112- M12000, 3231+06
113- MTRX-121, 7474+07, 3805+07, 1206+07, 8250-01, 1820+00, 1828+00, 9418+06, 6612+06+M12100
114- M12100, 7894+06
115- MTRX-122, 7358+07, 3652+07, 1363+07, 1039+00, 1686+00, 1696+00, 8368+06, 6545+06+M12200
116- M12200, 7100+06
117- MTRX-123, 7211+07, 1654+07, 1654+07, 1437+00, 1437+00, 1451+00, 6421+06, 6421+06+M12300
118- M12300, 5626+06
119- MTRX-124, 7132+07, 8015+06, 1811+07, 1652+00, 1303+00, 1319+00, 5372+06, 6354+06+M12400
120- M12400, 4832+06
121- MTRX-125, 3362+07, 3781+07, 1181+07, 7853-01, 1780+00, 1779+00, 8337+06, 5533+06+M12500
122- M12500, 7804+06
123- MTRX-126, 3362+07, 3107+07, 1855+07, 1043+00, 1522+00, 1522+00, 7610+06, 6260+06+M12600
124- M12600, 7804+06
125- MTRX-127, 3362+07, 1855+07, 3107+07, 1522+00, 1043+00, 1047+00, 6260+06, 7610+06+M12700
126- M12700, 7804+06
127- MTRX-128, 3362+07, 1181+07, 3781+07, 1780+00, 7853-01, 7907-01, 5533+06, 8337+06+M12800
128- M12800, 7804+06
129- MTRX-129, 1788+07, 3767+07, 1168+07, 7639-01, 1759+00, 1752+00, 7756+06, 4951+06+M12900
130- M12900, 7756+06
131- MTRX-130, 1211+07, 3136+07, 2120+07, 1045+00, 1434+00, 1429+00, 7202+06, 6106+06+M13000
132- M13000, 8183+06
133- MTRX-131, 1290+07, 1963+07, 3889+07, 1568+00, 8309-01, 8292-01, 6173+06, 8250+06+M13100
134- M13100, 8077+06
135- MTRX-132, 1332+07, 1332+07, 4841+07, 1849+00, 5063-01, 5063-01, 5619+06, 9405+06+M13200
136- M13200, 9405+06
137- MTRX-133, 2504+08, 1919+07, 1009+07, 1478+00, 1826+00, 1883+00, 1110+07, 1012+07+M13300
138- M13300, 5227+06

BULK DATA CARDS

FIELD=1, FIELD=2, FIELD=3, FIELD=4, FIELD=5, FIELD=6, FIELD=7, FIELD=8, FIELD=9, FIELD=10

WTX-130, 2500+06, 1640+07, 9664+06, 1545+00, 1803+00, 1866+00, 1067+07, 9943+06, 13400
 +13400, 4800+06
 WTX-135, 2492+04, 1123+07, 8875+06, 1669+00, 1759+00, 1818+00, 9876+06, 9622+06, 13500
 +13500, 4006+06
 WTX-136, 2488+04, 8450+06, 8450+06, 1736+00, 1736+00, 1795+00, 9448+06, 9448+06, 13600
 +13600, 3578+06
 WTX-137, 1677+04, 1881+07, 9713+06, 1417+00, 1765+00, 1807+00, 9477+06, 8456+06, 13700
 +13700, 5089+06
 WTX-138, 1677+04, 1645+07, 1207+07, 1507+00, 1675+00, 1717+00, 9183+06, 8710+06, 13800
 +13800, 5089+06
 WTX-139, 1677+04, 1207+07, 1645+07, 1675+00, 1507+00, 1550+00, 8710+06, 9183+06, 13900
 +13900, 5089+06
 WTX-140, 1677+04, 9713+06, 1881+07, 1765+00, 1417+00, 1461+00, 8456+06, 9437+06, 14000
 +14000, 5089+06
 WTX-141, 7132+07, 1811+07, 9015+06, 1303+00, 1652+06, 1665+00, 6354+06, 5372+06, 14100
 +14100, 4832+06
 WTX-142, 7211+07, 1654+07, 1654+07, 1437+00, 1437+00, 1451+00, 6421+06, 6421+06, 14200
 +14200, 5626+06
 WTX-143, 7358+07, 1363+07, 3053+07, 1686+00, 1039+00, 1054+00, 6545+06, 8368+06, 14300
 +14300, 7100+06
 WTX-144, 7436+07, 1206+07, 3803+07, 1820+00, 8250+01, 8404+01, 6612+06, 9416+06, 14400
 +14400, 7894+06
 WTX-145, 8639+06, 1774+07, 6639+06, 1242+00, 1590+00, 1588+00, 4693+06, 3712+06, 14500
 +14500, 4693+06
 WTX-146, 9852+06, 1659+07, 1895+07, 1400+00, 1309+00, 1308+00, 4933+06, 5188+06, 14600
 +14600, 5915+06
 WTX-147, 1211+07, 1446+07, 3810+07, 1692+00, 7874+01, 7868+01, 5379+06, 7929+06, 14700
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CARD

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BULK DATA CARDS

FIELD=1, FIELD=2, FIELD=3, FIELD=4, FIELD=5, FIELD=6, FIELD=7, FIELD=8, FIELD=9, FIELD=10

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234-	WTX-234, 1426-05, 4298-05, 4860-05								
235-	WTX-235, 1168-05, 4405-05, 4601-05								
236-	WTX-236, 1026-05, 4462-05, 4462-05								
237-	WTX-237, 1947-05, 3732-05, 4490-05								
238-	WTX-238, 1947-05, 3929-05, 4294-05								
239-	WTX-239, 1947-05, 4294-05, 3929-05								
240-	WTX-240, 1947-05, 4490-05, 3732-05								
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270-	SOC1 13 1 3								
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H U L K D A T A C A R D S

FIELD-1, FIELD-2, FIELD-3, FIELD-4, FIELD-5, FIELD-6, FIELD-7, FIELD-8, FIELD-9, FIELD-10

CARD

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281-	23	1		6			.0124-2		
282-	23	1			8		.05-2		
283-	22	1	1	1	5	6	2		
284-	22	1	2	1	4	8	5		
285-	22	1	3	1	2	3	4		
286-	22	1	1	4	8	7	3		
287-	22	1	3	5	6	7	8		
288-	22	1		8			.05-2		
289-	22	1							
290-	22	1							

END DATA

Cracked Bar
Bulk Data Model

PAGE 1

BULK DATA CARDS

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3	4	4	9	5						+A2
4										
5	2	1	9	10	6					+C2
6	2	5	9	3						
7	1	0	10	6						
8	1	0	0	0						
9	2	1.0	0	0						
10	3	1.77	0	0						
11	4	0	0	1.48						
12	5	0	0	1.48						
13	6	1.77	0	1.48						
14	1	1.77	0	1.48						
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16	1				0	30				
17	2	1.7								
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PARAM EREFL 1,F-R
END DATA

ATJ-S Bar Specimen
Bulk Data Model

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R U L K D A T A C A R D S

FIELD=1, FIELD=2, FIELD=3, FIELD=4, FIELD=5, FIELD=6, FIELD=7, FIELD=8, FIELD=9, FIELD=10

CARD	1	2	3	4	5	6	7	8	9	10
1- CPD3	1	1	1	5						
2- +C	4	4	4	6						
3- GRID	1	0.0	0.0	0.0	0.0					
4- GRID	2	0.125	0.0	0.0	0.0					
5- GRID	3	0.125	0.0	0.0	0.125					
6- GRID	4	0.0	0.0	0.0	0.125					
7- PATCH	1	1	2	3	4					
8- MP	1	1	1	1						
9- PATCH	1	1	1	1						
10- MTRX=1	1.8+5	1.8+5	1.2+6	0.1	0.15	0.15	0.0	30.0	3	
11- +M	6.5+5						0.15	8.0+5	6.5+5	+M
12- CPD3	1	1	1	1	1	1	1	2		
13- SDC10	10	1	3	1	1	1	5	4		
14- SDC20	10	1	3	1	1	1	3	4		
15- SDC20	10	1	3	1	1	1	6	4		
16- DATAG	1	1.0	2	1.0	3	1.0	1.0	4	1.0	+D
17- +D	5		6	3	6	3	5			
18- DPAT	10	1	2	3	6	6	5			
19- DPAT	20	1	4	4	4	4	5			
20- FLEAD3	10	1	10	-4.0+3	6	6	5			
21- FLEAD3	10	1	20	-4.0+3	6	6	5			
22- END DATA										

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are characterized in terms of those state variables that correlate a materials response such as effective stress or strain energy. A parametric cubic representation is used for all state variables, the solid geometry and all physical properties. The associated finite element extends isoparametric modeling to allow properties, linear or nonlinear, to vary over the volume of an element as in rosette material construction. Applications of the model to a carbon-carbon unit cell, to strain singularities and to the inelastic response of a graphite bar illustrate its utility. Good agreement with triaxial test data for inelastic strains under hydrostatic pressure is obtained. Numerical results are computed using PATCHES-III and the conjugate gradient algorithm without the assembly of large matrices. This approach is tailored for vector processors and can reduce the high cost of nonlinear three-dimensional analyses.

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